

## Synchrotron powder X-ray diffraction study of the structure and dehydration behavior of sepiolite

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### ABSTRACT

Rietveld refinements using synchrotron powder X-ray diffraction data were used to study the crystal structure and dehydration behavior of sepiolite from Durango, Mexico. The room-temperature (RT) sepiolite structure in air compares well with previous models but reveals an additional zeolitic H<sub>2</sub>O site. The RT structure under vacuum retained only ~1/8 of the zeolitic H<sub>2</sub>O and the volume decreased by 1.3%. Real-time, temperature-resolved synchrotron powder X-ray diffraction data and Rietveld refinements were used to investigate the behavior of the sepiolite structure from 300 to 925 K. Rietveld refinements revealed that most of the zeolitic H<sub>2</sub>O is lost by ~390 K, accompanied by a decrease in the *a* and *c* unit-cell parameters. Above ~600 K the sepiolite structure folds as one-half of the crystallographically bound H<sub>2</sub>O is lost. Rietveld refinements of the “anhydrous” sepiolite structure reveal that, in general, unit-cell parameters *a* and *b* and volume steadily decrease with increasing temperature; there is an obvious change in slope at ~820 K suggesting a phase transformation coinciding with the loss of the remaining bound H<sub>2</sub>O molecule.

**Keywords:** Sepiolite, Rietveld, synchrotron, powder diffraction

### INTRODUCTION

Sepiolite is a hydrous Mg-rich silicate clay mineral with fibrous morphology that typically occurs as fine-grained, poorly crystalline masses. It occurs in a wide variety of geological environments and has been mined for centuries because of its many useful properties. Sepiolite boasts more than 100 commercial uses today in the pharmaceutical, fertilizer, and pesticide industries (Van Scoyoc et al. 1979; Jones and Galan 1988), and its versatile functionality derives from the large surface area and microporosity that are characteristic of the material. In recent years, sepiolite has received considerable attention with regard to the adsorption of organics, for use as a support for catalysts (Jones and Galan 1988), as a molecular sieve (Ruiz-Hitzky 2001), and as an inorganic membrane for ultrafiltration (Wang et al. 2001).

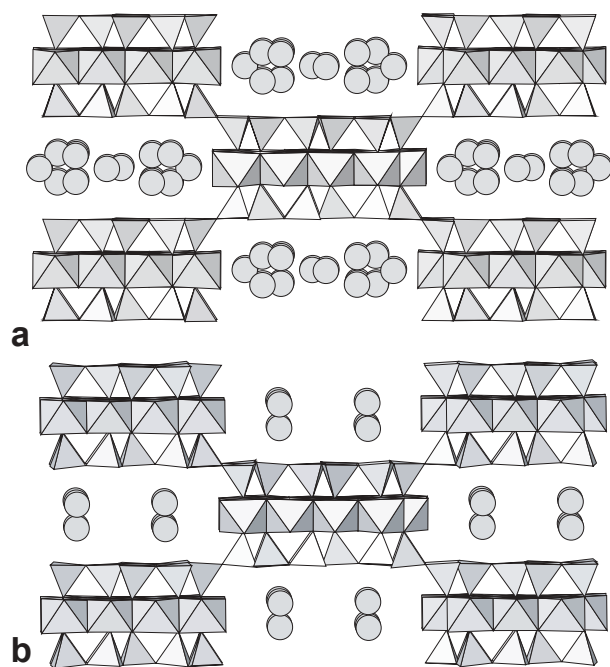
Sepiolite has a layer structure constructed of ribbons of 2:1 phyllosilicate modules that are linked triple silicate chains that sandwich Mg-(O,OH) octahedral strips (Fig. 1). The ribbons are linked by inversion of the SiO<sub>4</sub> tetrahedra through Si-O-Si bonds. Sepiolite has continuous tetrahedral sheets, but with tetrahedral apices pointing in opposite directions in adjacent ribbons. Unlike other phyllosilicates, it lacks continuous octahedral sheets. The structure has large tunnels parallel to the phyllosilicate ribbons that are partially occupied by H<sub>2</sub>O molecules. The tunnels measure 3.7 × 10.6 Å in cross section.

The sepiolite structure was first deduced by Brauner and

Preisinger (1956) from fiber X-ray diffraction patterns, and their model is orthorhombic with space group *Pnan*. Rautureau and Tchoubar (1976) calculated projected electron-density maps using electron-diffraction intensities that confirmed the model of Brauner and Preisinger (1956) but suggested the possibility of a split site for one of the Mg atoms and crystallographically bound H<sub>2</sub>O molecules. Because of its fine-grained and poorly crystalline nature, it has not been possible to study the crystal structure of sepiolite using single-crystal diffraction methods, and consequently many details of the structure are still not well known. In this study, we used synchrotron powder X-ray diffraction data and Rietveld refinements to better characterize the structure and behavior of sepiolite. We refined room-temperature structures for both a fully hydrated sepiolite and one for which the zeolitic H<sub>2</sub>O molecules were removed under vacuum.

In the second part of our study, we used temperature-resolved real-time synchrotron X-ray powder diffraction and Rietveld refinements to investigate in unprecedented detail changes in the sepiolite structure as it dehydrated from room temperature (RT) to 925 K. The Brauner and Preisinger (1956) structure model includes three types of H<sub>2</sub>O: zeolitic H<sub>2</sub>O molecules in the tunnels, crystallographically bound H<sub>2</sub>O molecules that complete the coordination of the Mg atoms at the edges of the octahedral strips, and hydroxyl (OH) groups bonded to some of the Mg atoms. Numerous previous studies have monitored the dehydration of sepiolite using thermogravimetric analysis (TGA) and differential thermal analysis (DTA) methods (e.g., Preisinger 1959; Hayashi 1969; Nagata et al. 1974; Serna et al. 1975; Lokanatha et al. 1985; Ruiz et al. 1996; Weir et al. 2002).

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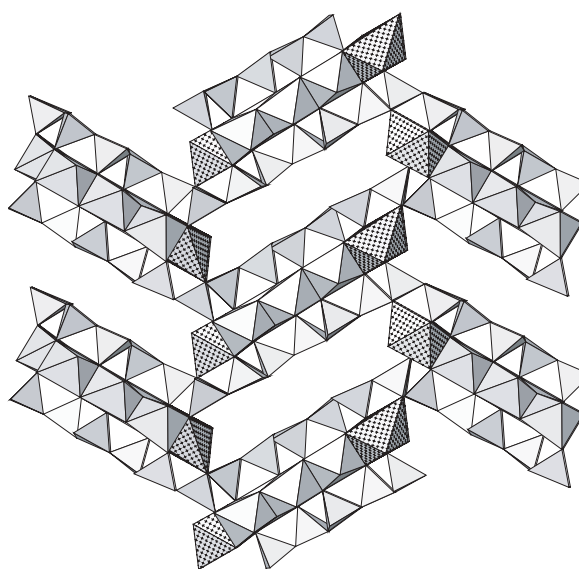


**FIGURE 1.** Polyhedral crystal structure representations, viewed along  $c$ , resulting from the Rietveld refinements for sepiolite (RT): (a) in air and (b) under vacuum. Shaded circles represent zeolitic  $\text{H}_2\text{O}$  molecules.

There is a general concurrence among these researchers that zeolitic  $\text{H}_2\text{O}$  is lost during heating from RT to  $\sim 415$  K, or at RT under vacuum; bound  $\text{H}_2\text{O}$  molecules are released in two stages between  $\sim 525$  and  $875$  K; and hydroxyl groups are lost above  $\sim 1075$  K. Preisinger (1959, 1963) proposed that loss of the bound  $\text{H}_2\text{O}$  causes a phase change during which the structure folds by rotation of the phyllosilicate ribbons about an axis through the Si-O-Si corner bonds that link the ribbons (Fig. 2). Rautureau and Mifsud (1977) show high-resolution transmission electron microscope images that appear to confirm the existence of a folded “anhydrous” structure. Furthermore, the TGA results and an X-ray diffraction study by Nagata et al. (1974) suggest that there are two distinct “anhydrous” sepiolite phases, corresponding to the loss of the first and second bound  $\text{H}_2\text{O}$  molecules, respectively. We present here the results of a series of refinements of the “anhydrous” sepiolite structure over the temperature range  $\sim 666$  to  $925$  °C that further confirm both the basic Preisinger (1959) folded-structure model and the existence of two “anhydrous” phases.

#### DATA COLLECTION AND REFINEMENT

The sepiolite sample used for this study (NMNH no. 92158-1) is from Durango, Mexico, and powder X-ray data revealed that it is exceptionally well ordered for sepiolite and free of impurity phases. Electron microprobe analyses (JEOL JXA-8900R) using samples embedded in epoxy, polished and carbon-coated yielded a result that is similar to the ideal chemical formula:  $\text{Mg}_{16}\text{Si}_{24}\text{O}_{60}(\text{OH})_8 \cdot n\text{H}_2\text{O}$ . Because of the difficulty of analyzing fibrous, embedded samples, it was not possible to obtain water by difference. Thermogravimetric analysis on 11.8 mg of sample (Universal 3.3B TA Instruments), heating at 5 K/min to 1273 K yielded a total weight loss of  $\sim 18\%$ , which is similar to results from previous studies (e.g., Preisinger 1959; Nagata et al. 1974). The sample used for X-ray diffraction was hand ground under acetone in an agate mortar and passed through a 400-mesh sieve. For the synchrotron X-ray diffraction (XRD) studies, samples were loaded



**FIGURE 2.** Folded structure resulting from the Rietveld refinement for sepiol- $2\text{H}_2\text{O}$  (742 K) viewed along the monoclinic  $c$  axis. The dotted polyhedra at the edges of the octahedral strips represent the  $\text{Mg}_4$  coordination.

into 0.7 and 0.5 mm quartz-glass capillaries for the RT and heating experiments, respectively. X-ray diffraction data were collected at beam line X7B of the National Synchrotron Light Source (NSLS), Brookhaven National Laboratory (BNL), using wavelengths of 0.9492 Å (RT samples in air and vacuum) and 0.9274 Å (heating experiment) and a MAR345 full imaging plate detector. Diffraction data used for the sepiolite structure refinements were collected at room temperature (300 K) from a sample in air and from a second sample under vacuum ( $1 \times 10^{-6}$  torr).

The heating experiment was performed in air using a Blake Instruments furnace with a Pt-13%Rh coiled wire yoke encased in  $\text{ZrO}_2$  cement (Brown et al. 1973). The temperature was varied with an Omega controller and monitored with a Chromel-Alumel thermocouple located  $\sim 2$  mm from the specimen. The actual sample temperature was determined for the range 298 to 1273 K by a variety of phase and melting transitions and by the placement of an additional thermocouple in the sample position. The highly linear relationship between the observed and actual temperatures ( $r^2 = 0.983$ ) allowed us to calculate a calibration curve with an estimated error of  $\pm 5$  K for a given temperature. Temperature-resolved data from 300 to 925 K were collected as a series of 90 s exposures with a MAR345 full imaging plate detector. The temperature was increased continuously at 5.4 K/min and measurements were obtained every  $\sim 15$  K, owing to down time for repositioning of the sample and reading the imaging plate; thus, each exposure encompassed a temperature range of  $\sim 8.1$  K. During each exposure the sample was rotated through a  $90^\circ$  angle. Preferred orientation of the powder was eliminated through a combination of the specimen rotation, use of a capillary sample holder, and full intensity integration of the diffraction rings, as obtained using the program Fit2D (Hammersley et al. 1996) with a polarization factor of 0.93.

Rietveld refinements were performed using the General Structure Analysis System (GSAS) of Larson and Von Dreele (2001) using data measured over a range of temperatures from RT (in air and vacuum) to the point at which most of the bound  $\text{H}_2\text{O}$  was lost (925 K). The starting structural parameters for the sepiolite phyllosilicate modules were taken from Brauner and Preisinger (1956). The model for the “anhydrous” sepiolite was derived graphically from that illustrated by Preisinger (1959, 1963), and Mg-O and Si-O bond distances were optimized in a monoclinic unit cell ( $P2_1/n$ ) using the GSAS bond distance constraint feature with a large weighting factor ( $F = 5000$ ). Initially, the Si-O distances were constrained to 1.60 Å and the Mg-O distances to 2.03 Å. The diffraction pattern backgrounds were fit using a linear interpolation function. Peak profiles were modeled by a pseudo-Voigt profile function as parameterized by Thompson et al. (1987) with asymmetry corrections by Finger et al. (1994) and microstrain anisotropic broadening terms by Stephens (1999). Displacement parameters for a given atom type were constrained to be equivalent.

**TABLE 1.** Final Rietveld refinement parameters for sepiolite in air and under vacuum

	Air	Vacuum
Space group	<i>Pncn</i>	<i>Pncn</i>
Unit cell		
<i>a</i> (Å)	13.405(1)	13.248(1)
<i>b</i> (Å)	27.016(1)	27.053(1)
<i>c</i> (Å)	5.2750(1)	5.2616(2)
<i>V</i> (Å <sup>3</sup> )	1910.4(1)	1885.7(2)
Refinement		
No. of data points	1871	2185
No. of reflections	1058	750
Diffraction range (d Å)	0.986–9.1	1.13–8.9
No. of variables	82	85
<i>R</i> ( <i>F</i> <sup>2</sup> )	0.055	0.052
<i>R</i> <sub>wp</sub>	0.021	0.020
$\chi^2$	0.125	4.91

**TABLE 2.** Final Rietveld refinement parameters for sepiol-2H<sub>2</sub>O (742 K) and anhydrous sepiolite (925 K)

	742 K	925 K
Space group	<i>P2<sub>1</sub>/n</i>	<i>P2<sub>1</sub>/n</i>
Unit cell		
<i>a</i> (Å)	23.446(4)	23.415(3)
<i>b</i> (Å)	11.352(3)	11.157(2)
<i>c</i> (Å)	5.2782(4)	5.2816(5)
$\beta$ (°)	89.06(1)	89.21(1)
<i>V</i> (Å <sup>3</sup> )	1407.7(5)	1379.7(5)
Refinement		
No. of data points	2023	2023
No. of reflections	1103	1065
Diffraction range (d Å)	1.13–6.6	1.13–6.6
No. of variables	121	121
<i>R</i> ( <i>F</i> <sup>2</sup> )	0.010	0.013
<i>R</i> <sub>wp</sub>	0.006	0.007
$\chi^2$	0.48	0.54

During the initial cycles of refinement, only the background, scale, peak profile, and unit-cell parameters were allowed to vary. After convergence, difference-Fourier maps were calculated to locate the O atoms of the H<sub>2</sub>O molecules in the tunnels (for the sample in air). These O atoms were added to the structure model and all atom positions and displacement parameters and occupancy factors of the O atoms of the H<sub>2</sub>O molecules were refined. Soft constraints were used to limit the Si–O bond distances to within the range observed in similar silicate structures, e.g., tupsertsuaitaite (1.58–1.63 Å, <1.61 Å>; Cámara et al. 2002) and raite (1.59–1.65 Å <1.62 Å>; Pluth et al. 1997). The weighting factors for the constraints were gradually reduced. Attempts to eliminate the weighting factors completely for sepiolite resulted in Si–O distances of 1.55 (Si1–O1) and 1.68 Å (Si2–O6), and therefore the smallest weighting factor (*F* = 0.5) that yielded reasonable Si–O distances (i.e., 1.60–1.64 Å) was retained. The bond constraint  $\chi^2$  contribution accounted for 24.2 out of a total  $\chi^2$  contribution of 224. For “anhydrous” sepiolite, the Si–O and Mg–O distances, except for those between Mg4 and O10, were constrained throughout the refinement (*F* = 50).

The final refinement parameters for RT sepiolite in air and vacuum are listed in Table 1, and results for “anhydrous” sepiolite (742 K) and anhydrous sepiolite (925 K) are included in Table 2. The refined atom positions for the samples in air and under vacuum at RT are not significantly different except for the zeolitic H<sub>2</sub>O; therefore, the complete set of atom positions is given in Table 3 for the sample in air, and only those for the remaining zeolitic H<sub>2</sub>O molecule are included for the sample under vacuum (The full set of atom positions for sepiolite under vacuum are listed in Appendix 1.)<sup>1</sup> Refined framework atom positions for “anhydrous”

**TABLE 3.** Atomic coordinates and isotropic displacement parameters for RT sepiolite (air) and for the zeolitic H<sub>2</sub>O site (vacuum)

Atom	<i>x</i>	<i>y</i>	<i>z</i>	Site occupancy factor	<i>U</i> <sub>iso</sub> (Å <sup>2</sup> )*
Mg1	0	0.0297(7)	0.25	1.0	0.040(2)
Mg2	0	0.0850(7)	0.75	1.0	0.040(2)
Mg3	0	0.1434(5)	0.25	1.0	0.040(2)
Mg4	0	0.2049(4)	0.75	1.0	0.040(2)
Si1	0.2026(6)	0.0282(4)	0.586(2)	1.0	0.040(1)
Si2	0.1973(8)	0.1409(3)	0.576(2)	1.0	0.040(1)
Si3	0.2042(7)	0.1949(2)	0.070(2)	1.0	0.040(1)
O1	0.0824(7)	0.0276(7)	0.585(2)	1.0	0.035(1)
O2	0.0805(9)	0.0854(6)	0.081(2)	1.0	0.035(1)
O3	0.0762(8)	0.1431(5)	0.597(2)	1.0	0.035(1)
O4	0.0847(8)	0.1971(6)	0.077(2)	1.0	0.035(1)
O5	0.248(1)	0.0040(7)	0.324(2)	1.0	0.035(1)
O6	0.2451(9)	0.0849(4)	0.579(3)	1.0	0.035(1)
O7	0.243(1)	0.1708(7)	0.334(2)	1.0	0.035(1)
O8	0.249(1)	0.1736(7)	0.802(2)	1.0	0.035(1)
O9	0.25	0.25	0.089(6)	1.0	0.035(1)
O10 (H <sub>2</sub> O)	0.088(1)	0.2596(5)	0.594(3)	1.0	0.035(1)
O11 (H <sub>2</sub> O)	0.562(1)	−0.090(1)	−0.012(4)	0.91(2)	0.10
O12 (H <sub>2</sub> O)	0.50	0.180(1)	0.25	1.0(3)	0.10
O13 (H <sub>2</sub> O)	0.50	0.023(3)	0.25	0.43(3)	0.10
O14 (H <sub>2</sub> O)	0.439(3)	0.157(3)	0.74(2)	0.33(3)	0.10
Vacuum					
O11 (H <sub>2</sub> O)	0.562(6)	0.109(3)	0.50(1)	0.24(2)	0.10

\* Isotropic displacement parameters for same atom types were constrained to be equal, and for O11–O14 were set equal to 0.10.

**TABLE 4.** Atomic coordinates and isotropic displacement parameters for Sepiol-2H<sub>2</sub>O (742 K)

Atom	<i>x</i>	<i>y</i>	<i>z</i>	Site occupancy factor	<i>U</i> <sub>iso</sub> (Å <sup>2</sup> )*
Mg1	0.0238(5)	0.023(1)	0.244(2)	1.0	0.02
Mg2	0.0862(6)	0.115(1)	0.729(2)	1.0	0.02
Mg3	0.1431(6)	0.179(1)	0.244(2)	1.0	0.02
Mg4	0.1942(5)	0.246(1)	0.719(2)	1.0	0.02
Si1	0.0869(6)	0.833(1)	0.561(2)	1.0	0.02
Si2	0.2012(5)	0.970(1)	0.539(2)	1.0	0.02
Si3	0.2504(6)	0.027(1)	0.028(2)	1.0	0.02
O1	0.0558(5)	0.959(1)	0.580(2)	1.0	0.025
O2	0.1024(5)	0.042(1)	0.074(2)	1.0	0.025
O3	0.1664(7)	0.0903(1)	0.568(3)	1.0	0.025
O4	0.2185(7)	0.153(2)	0.040(3)	1.0	0.025
O5	0.0680(9)	0.744(2)	0.340(3)	1.0	0.025
O6	0.1529(7)	0.870(2)	0.574(4)	1.0	0.025
O7	0.235(1)	0.950(2)	0.273(3)	1.0	0.025
O8	0.2387(9)	0.947(2)	0.784(3)	1.0	0.025
O9	0.3172(7)	0.034(2)	0.061(4)	1.0	0.025
O10 (H <sub>2</sub> O)	0.2733(9)	0.211(3)	0.512(3)	1.0(2)	0.025
Si1b	0.9668(5)	0.234(1)	0.916(2)	1.0	0.02
Si2b	0.0807(5)	0.374(1)	0.894(2)	1.0	0.02
Si3b	0.1278(6)	0.454(1)	0.405(2)	1.0	0.02
O1b	0.0092(6)	0.124(1)	0.923(3)	1.0	0.02
O2b	0.0601(6)	0.174(2)	0.386(2)	1.0	0.025
O3b	0.1172(6)	0.257(1)	0.917(2)	1.0	0.025
O4b	0.1653(9)	0.336(1)	0.407(3)	1.0	0.025
O5b	0.9260(9)	0.227(2)	0.166(3)	1.0	0.025
O6b	0.0152(8)	0.332(2)	0.868(4)	1.0	0.025
O7b	0.092(1)	0.446(2)	0.152(3)	1.0	0.025
O8b	0.0798(8)	0.443(2)	0.627(3)	1.0	0.025

\* Isotropic displacement parameters were fixed to typical values for silicate structures.

<sup>1</sup> Deposit Item AM-07-001, Appendix 1, the full set of atom positions for sepiolite under vacuum. Deposit items are available two ways: For paper copy contact the Business Office of the Mineralogical Society of America (see inside cover of recent issue) for price information. For an electronic copy visit the MSA web site at <http://www.minsocam.org>, go to the American Mineralogist Contents, find the table of contents for the specific volume/issue wanted, and then click on the deposit link there.

sepiolite were similar at all temperatures and therefore only the values determined at 742 K are reported in Table 4. Selected bond distances for RT sepiolite in air and “anhydrous” sepiolite (742 K) are reported in Tables 5 and 6, respectively. The final observed, calculated, and difference patterns for the RT sepiolite sample in air and “anhydrous” sepiolite (742 K) are plotted in Figures 3 and 4, respectively. As is often the case, the standard deviations calculated by GSAS for the unit-cell parameters are likely lower than the true errors (Post and Bish 1989). We present the errors calculated by GSAS in the tables of this paper with the understanding that the actual errors may be more than an order of magnitude greater than the calculated deviations.

**TABLE 5.** Selected bond distances (Å) for sepiolite in air at RT

Mg1-O1	2.09(1) × 2	Mg2-O1	2.09(1) × 2
Mg1-O1	2.09(1) × 2	Mg2-O2	2.05(1) × 2
Mg1-O2	2.06(1) × 2	Mg2-O3	2.04(1) × 2
<Mg1-O>	2.08	<Mg2-O>	2.06
Mg3-O2	2.10(1) × 2	Mg4-O3	2.12(1) × 2
Mg3-O3	2.10(1) × 2	Mg4-O4	2.08(1) × 2
Mg3-O4	2.06(1) × 2	Mg4-O10	2.06(1) × 2
<Mg3-O>	2.09	<Mg4-O>	2.09
Si1-O1	1.611(6)	Si2-O3	1.627(6)
Si1-O5	1.642(6)	Si2-O6	1.643(6)
Si1-O5	1.644(6)	Si2-O7	1.631(6)
Si1-O6	1.635(6)	Si2-O8	1.639(6)
<Si1-O>	1.63	<Si2-O>	1.64
Si3-O4	1.604(6)	O10-O8	3.13(2)
Si3-O7	1.627(7)	O10-O10	2.93(3)
Si3-O8	1.639(7)	O10-O12	2.72(2)
Si3-O9	1.614(6)	O10-O14	2.47(8)
<Si3-O>	1.62	O12-O14	2.78(8)
O11-O11	3.01(4)	O12-O14	2.88(8)
O11-O12	2.87(3)	O13-O13	2.91(6)
O11-O13	2.35(6)	O14-O8	2.64(7)
O11-O14	2.32(8)		
O11-O14	2.79(8)		
O11-O14	2.79(8)		

Note: Soft constraints were applied to Si-O distances (see text) during refinement.

**TABLE 6.** Selected bond distances (Å) for sepiol-2H<sub>2</sub>O (742 K)

Mg1-O1	2.07(1)	Mg2-O1	2.07(1)
Mg1-O1	2.08(1)	Mg2-O2	2.04(1)
Mg1-O2	2.05(1)	Mg2-O3	2.07(1)
Mg1-O1b	2.08(1)	Mg2-O1b	2.07(1)
Mg1-O1b	2.05(1)	Mg2-O2b	2.04(1)
Mg1-O2b	2.06(1)	Mg2-O3b	2.03(1)
<Mg1-O>	2.07	<Mg2-O>	2.05
Mg3-O2	2.04(1)	Mg4-O3	2.05(1)
Mg3-O3	2.07(1)	Mg4-O4	2.09(1)
Mg3-O4	2.08(1)	Mg4-O8	2.77(2)
Mg3-O2b	2.07(1)	Mg4-O10	2.18(3)
Mg3-O3b	2.04(1)	Mg4-O3b	2.07(1)
Mg3-O4b	2.05(1)	Mg4-O4b	2.06(1)
<Mg3-O>	2.06	<Mg4-O>	2.20
O10-O3	2.87(3)	O10-O4	2.90(3)
O10-O8	2.89(4)	O10-O4b	2.96(3)

Note: Soft constraints were applied to Si-O and Mg-O distances (except Mg4-O10) during refinements.

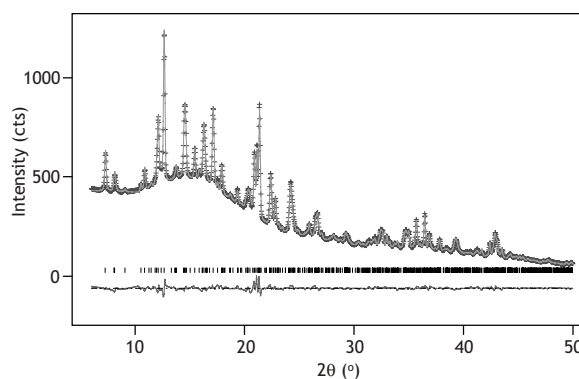
## RESULTS AND DISCUSSION

### Sepiolite at RT in air

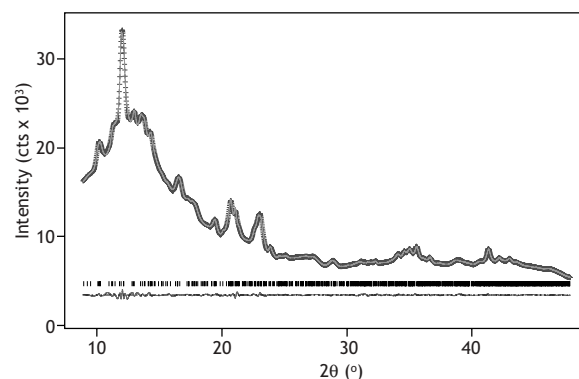
In general, the refined sepiolite model determined using the XRD data collected at room temperature in air (Fig. 1) compares well with that deduced by Brauner and Preisinger (1956). In addition to minor differences in some of the atom positions, the refinement revealed an additional zeolitic H<sub>2</sub>O site.

The refined Mg-O distances range from 2.04 to 2.12 Å (Table 5), consistent with relatively undistorted Mg-O octahedra. The mean Mg-O distances for the four unique octahedra (2.06–2.09 Å) are close to the ideal value of 2.08 Å (Shannon 1976), which is consistent with the microprobe analyses indicating that only Mg resides in the octahedral sites. As discussed above, the Si-O distances were softly constrained during the refinement, and all fall in the range of 1.60–1.64 Å (Table 5).

The sepiolite model of Brauner and Preisinger (1956) contains three hydrous species: OH<sup>-</sup> anions at the O2 site, coordinated to Mg1, Mg 2, and Mg3; structural H<sub>2</sub>O at the O10 site, where it completes the Mg4 coordination at the edges of the Mg-(O, OH, H<sub>2</sub>O) octahedral strips; and three zeolitic H<sub>2</sub>O positions in the



**FIGURE 3.** Final observed (crosses), calculated (solid line), and difference (lower) patterns for the Rietveld refinement for sepiolite in air using the RT synchrotron powder X-ray diffraction data.



**FIGURE 4.** Final observed (crosses), calculated (solid line), and difference (lower) patterns for the Rietveld refinement for sepiol-2H<sub>2</sub>O in air using synchrotron powder X-ray diffraction data collected at ~742 K.

tunnels. Because of the negligibly small X-ray scattering factor for H, we were unable to locate the positions of the H atoms, and we assumed that O2 is OH<sup>-</sup> and O10 is structural H<sub>2</sub>O. The refined occupancy factor for the O10 site indicated that it is fully occupied. No evidence was found for split Mg4 and O10 sites as reported by Rautureau and Tchoubar (1976) from their electron diffraction study. These authors suggested that these split sites might be a feature of sepiolite in vacuum, as was the case for their sample in the TEM. However, no evidence was observed that Mg4 and O10 occupy split sites in the structure even for data collected from a sepiolite sample under vacuum.

Difference-Fourier maps and subsequent Rietveld refinement revealed four zeolitic H<sub>2</sub>O sites, three of which (O11, O12, and O13) are close to those in the Brauner and Preisinger (1956) model. Refined occupancy factors indicated that the O11 and O12 sites are at full occupancy, or nearly so, O13 is approximately one-half filled, and O14 (not reported by Brauner and Preisinger 1956) is about one-third occupied. The total of ~15.6 zeolitic H<sub>2</sub>O molecules per unit cell determined by refinement is in reasonable agreement (within errors of the occupancy factors) with the value of ~16 reported in the chemical formula determined by Preisinger (1959) from wet chemical analyses. The interatomic distances among the zeolitic H<sub>2</sub>O (O) sites range from 2.32 to 3.01 Å (Table 5) and considering the errors are typical for H-bonded distances.

The O14-O14 distance of 1.64 Å indicates that H<sub>2</sub>O molecules cannot occupy adjacent O14 sites and this is consistent with a refined occupancy for the site of less than one-half. The O12 and O14 sites are the only zeolitic H<sub>2</sub>O molecules within reasonable H-bonded distances, 2.72 and 2.47 Å, respectively, of the O10 (structural H<sub>2</sub>O); O14 is 2.64 Å from O8 and is the only zeolitic H<sub>2</sub>O molecule close to one of the framework O atoms. These results are similar to those of Giustetto and Chiari (2004) who recently reported a neutron diffraction study of palygorskite, which has a slightly smaller tunnel structure than that of sepiolite. They determined that the structural H<sub>2</sub>O in palygorskite that is comparable to O10 in sepiolite also forms apparent H-bonds with two of the zeolitic H<sub>2</sub>O molecules, at distances of ~2.55 and 2.85 Å. Thus, although the zeolitic H<sub>2</sub>O molecules in sepiolite exhibit a reasonable degree of H-bonding among themselves, they are not strongly linked to the surrounding structural framework, perhaps explaining why they are easily lost when sepiolite is exposed to vacuum or undergoes gentle heating.

### Sepiolite in vacuum

The refined atomic coordinates for the sepiolite framework under vacuum are not significantly different from those in air. The major difference between the two structures is loss of most of the zeolitic H<sub>2</sub>O from the sepiolite in vacuum (Fig. 1), and the consequent adjustment of the unit-cell parameters (Table 1). Difference-Fourier maps for the evacuated material show only a single electron-density peak in the tunnels (O11 in Table 3), near the O14 position of sepiolite in air (Table 3), and the refinement of the occupancy factor indicates that the site is approximately one-quarter filled (~2 H<sub>2</sub>O/cell). Therefore, sepiolite under vacuum, even at room temperature, releases ~7/8 of its zeolitic H<sub>2</sub>O. The three unit-cell edges decrease slightly with the H<sub>2</sub>O loss, the major change occurring along *a* (decrease of ~1.2%, vs. an increase of 0.1% for *b* and decrease of 0.2% for *c* (Table 1), resulting in a volume decrease of ~1.3%. Thus, the loss of the zeolitic H<sub>2</sub>O induces the sepiolite tunnels to collapse slightly along *a* and to relax along *b*.

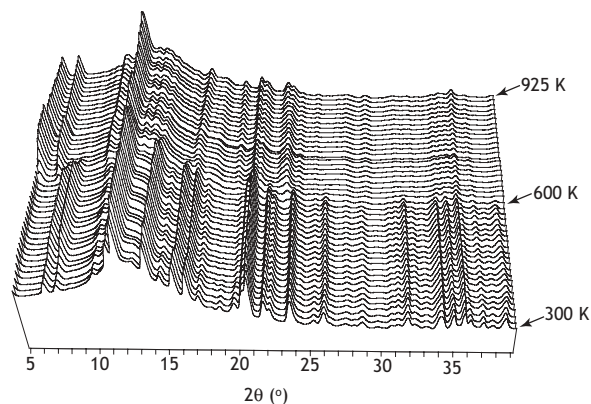
### In-situ heating experiment

The synchrotron X-ray diffraction data (Fig. 5) reveal an obvious phase change at ~600–650 K for sepiolite heated in air, and this transition presumably corresponds to the folding of the sepiolite structure as the first half of the structural H<sub>2</sub>O (O10) is lost (Preisinger 1959, 1963; Nagata et al. 1974). This sepiolite “anhydride” was designated as “sepiol-2H<sub>2</sub>O,” following the nomenclature of Kitayama et al. (1999), where “2H<sub>2</sub>O” refers to the two remaining structural H<sub>2</sub>O molecules, from the original four (based on a chemical formula normalized to 8 Mg atoms). Preisinger (1963) was the first to describe this dehydration product when sepiolite is heated above ~650 K, and he reported the phase as monoclinic (*P*2<sub>1</sub>/*n*).

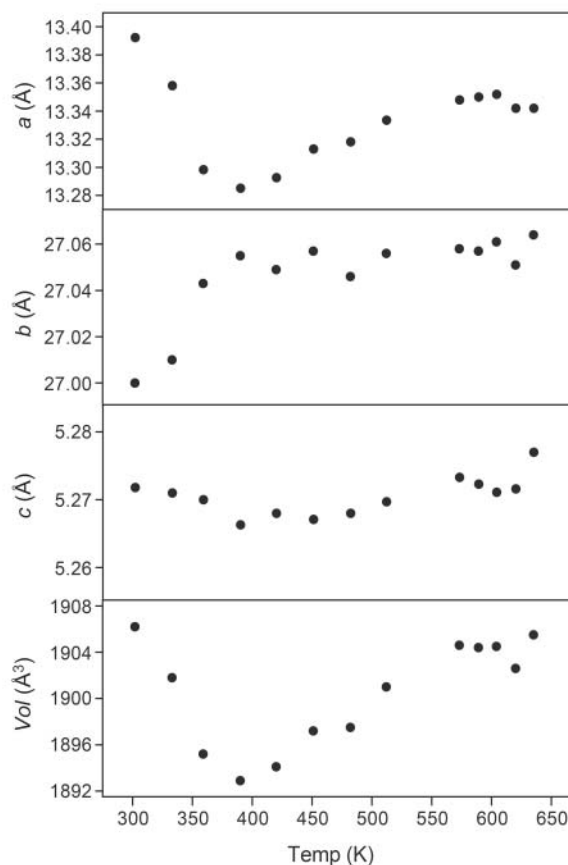
**RT to 600 K.** During the temperature-resolved synchrotron XRD heating of sepiolite in air, the unit-cell volume decreased continuously by ~0.7% between 300 and 390 K accompanied by a decrease in *a* and *c* and a slight increase in *b* (Fig. 6). Rietveld refinements using the XRD data indicated that most of the zeolitic H<sub>2</sub>O is lost over this same temperature range (Fig. 7). Similarly, Kitayama et al. (1999) observed a loss of zeolitic H<sub>2</sub>O

from sepiolite near 373 K, and TGA measurements of sepiolite by Nagata et al. (1974) indicated that H<sub>2</sub>O is lost continuously from just above RT to ~383 K. Serna et al. (1975) observed H<sub>2</sub>O loss in air from RT to ~450 K with a DTG peak at 383 K.

The Rietveld refinements indicated that as sepiolite is heated, H<sub>2</sub>O is lost at approximately the same rate from all of the zeo-



**FIGURE 5.** Synchrotron powder X-ray diffraction patterns for sepiolite in air vs. temperature, from 300 K (front pattern) to 925 K. The transformation to the folded sepiol-2H<sub>2</sub>O phase occurs in the range 600 to 650 K.



**FIGURE 6.** Plots of sepiolite unit-cell parameters vs. *T* for the range 300 to 600 K. The errors determined by the refinements are within the plotting symbols.

litic H<sub>2</sub>O sites. The occupancy refinements suggested that small amounts of H<sub>2</sub>O remain on the O14 and O11 sites to near 473 K. Note, however, that these small residual occupancies are near zero, within error; they might be artifacts resulting from correlations during the least-squares refinements among the occupancy factors, background coefficients and displacement parameters. In contrast, TGA studies (e.g., Serna et al. 1975) showed that zeolitic H<sub>2</sub>O loss might continue to at least ~450 K. Also, as discussed above, O14 probably forms H-bonds with O10 and O8, and therefore, some H<sub>2</sub>O molecules might linger at this site to higher temperatures. Above 390 K, there is a continual thermal expansion of the sepiolite structure to ~573 K (Fig. 6). The unit-cell volume increases by 0.6%, with most of the change occurring in *a* (+0.5%), probably corresponding to a slight inflation of the tunnel in that direction. Previous studies show that for sepiolite samples heated to less than ~625 K, zeolitic H<sub>2</sub>O is quickly replaced at ambient temperature and humidity (e.g., Preisinger 1959).

**600 to 925 K.** The powder XRD patterns in Figure 5 show the first appearance of sepiol-2H<sub>2</sub>O at ~600 K, as indicated by diffraction peaks at ~10.3 and 8.2 Å (~5° and 6° 2θ, respectively in Fig. 5), and over the next ~50 K the diffraction intensity for sepiol-2H<sub>2</sub>O increases as the sepiolite peaks diminish, until at ~650 K only sepiol-2H<sub>2</sub>O remains. Rietveld refinements of the sepiolite structure using diffraction patterns in this temperature range did not indicate any decrease in the O10 (structural H<sub>2</sub>O) occupancy prior to the phase change. This result suggests that the initial loss of O10 and consequent folding of the structure occur quickly, and the refined model reflects only the diminishing fraction of the sepiolite sample that retains all of its O10. This rapid transition is consistent with dehydration studies of Nagata et al. (1974) and Serna et al. (1975) that show narrow, intense DTA peaks associated with the first loss of structural H<sub>2</sub>O centered at ~573 and ~603 K, respectively. Serna et al. (1975) noted that the temperature range of the phase transition is dependent on the experimental conditions, such as the heating rate or whether the sample was heated in air or vacuum. Nonetheless, the transition temperature range observed in this study is consistent with those reported by previous researchers.

The refined model for sepiol-2H<sub>2</sub>O confirms that proposed by Preisinger (1959, 1963). The loss of one-half of the structural H<sub>2</sub>O causes the sepiolite phyllosilicate ribbons to rotate, or fold,

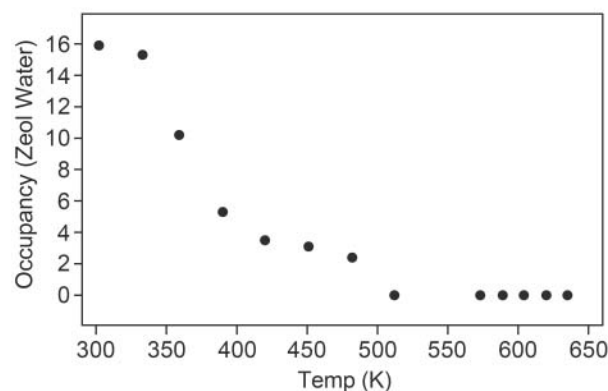


FIGURE 7. Plot of total number of zeolitic H<sub>2</sub>O molecules determined by Rietveld refinements vs. *T*.

thereby reducing the cross-sectional area of the tunnels and lowering the structure symmetry from orthorhombic to monoclinic. The unit-cell parameters for sepiol-2H<sub>2</sub>O at 742 K are listed in Table 2 and differ slightly from those proposed by Preisinger (1963); the *b* parameter is larger (11.35 Å rather than 10.9 Å), and β is ~89° instead of 90°.

The loss of one-half of the O10 H<sub>2</sub>O molecules requires Mg4 to complete its octahedral coordination in sepiol-2H<sub>2</sub>O by forming a bond, albeit a long one (2.76 Å), to one of the tetrahedral O atoms (O8) (Fig. 2). The Si-O bond constraints used during the refinements, however, might prevent complete refinement that could result in a shorter Mg4-O8 distance. The need to complete the Mg4 coordination octahedron after the loss of one of the O10 H<sub>2</sub>O molecules might be the major driving force to cause the sepiolite structure to fold, facilitated by having fewer O10 H<sub>2</sub>O molecules to prop open the tunnels.

Difference-Fourier maps calculated using only the sepiol-2H<sub>2</sub>O framework atoms revealed the position of the remaining O10 H<sub>2</sub>O molecule in the tunnel. The refined occupancy factor for O10 did not differ significantly from 1.0 for temperatures less than ~820 K. Above this temperature, the occupancy factor declined steadily (Fig. 8) to ~0.5 at 925 K, presumably as sepiol-2H<sub>2</sub>O transformed to “anhydrous” sepiolite (though the hydroxyl was retained in this high-temperature structure). Refinement using a second data set collected after heating at 925 K for an additional 12 minutes yielded an O10 occupancy factor of 0.4, suggesting that the heating rate was not yielding equilibrium loss of O10. In fact, DTA measurements by Nagata et al. (1974) show a considerably broader peak associated with the loss of the residual O10, compared with that for the initial O10, indicating a more gradual loss over a larger temperature range (673 to 893 K). Serna et al. (1975) argue that it is more difficult for the second half of the O10 water molecules to leave the structure because they are entrapped within the collapsed tunnels.

Lokanatha et al. (1985) described the loss of the second bound H<sub>2</sub>O molecule between 723 and 873 K, and their XRD radial distribution study revealed a decrease in the Mg4 coordination between 673 and 873 K. Bond distances calculated from our refined models indicated a lengthening of the Mg4-O10 distance from ~2.13 to 2.32 Å over the temperature range ~790 to 900 K. With the loss of all O10 H<sub>2</sub>O, Mg4 is coordinated to four O atoms at ~2.06 Å and O8 at 2.67 Å. Our diffraction studies and previous experiments (Lokanatha et al. 1985) show that as hydroxyl

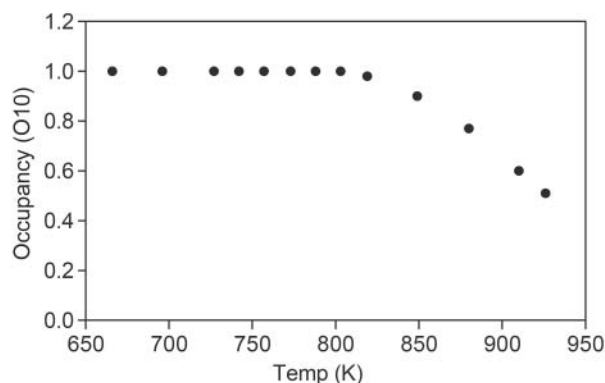
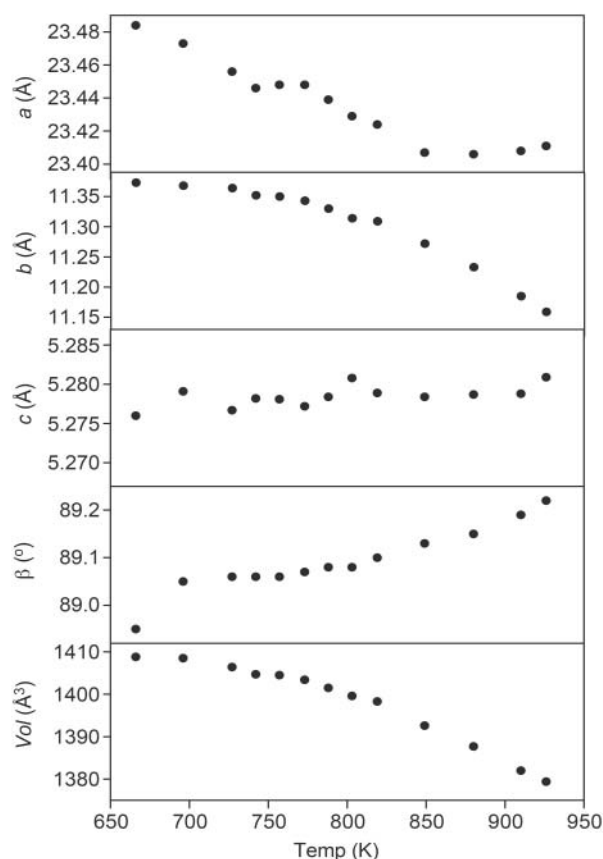


FIGURE 8. Plot of the refined occupancy factor for the bound H<sub>2</sub>O (O10) in sepiol-2H<sub>2</sub>O vs. *T*.



**FIGURE 9.** Plots of sepiol-2H<sub>2</sub>O and “anhydrous” sepiolite (>820 K) unit-cell parameters vs.  $T$  for the range 600 to 925 K. The errors determined by the refinements are within the plotting symbols.

groups are lost above ~1050 K an amorphous phase results.

The unit-cell parameters for the folded sepiolite structure are plotted as a function of temperature in Figure 9. In general,  $a$ ,  $b$ , and volume steadily decrease with increasing temperature and  $\beta$  approaches 90°; there is an obvious change in slope at ~820 K suggesting a phase transformation. The fact that the O10 occupancy factor also begins to decrease at about the same temperature (Fig. 8) strongly indicates that the phase change is caused by the loss of the second half of the bound H<sub>2</sub>O molecules (O10), i.e., sepiol-2H<sub>2</sub>O changes to “anhydrous” sepiolite (with OH). Nagata et al. (1974) similarly noted a discontinuous change in powder XRD peaks for samples heated between 623 and 723 K and suggested that there were two distinct dehydrated phases, one at 573 and another at 873 K. In contrast, Serna et al. (1975) did not observe any structural changes with the loss of the second bound H<sub>2</sub>O. Our results clearly suggest the existence of two distinct phases, one containing one-half of the bound H<sub>2</sub>O molecules and the other with no bound H<sub>2</sub>O. Above 820 K,  $b$ ,  $\beta$ , and volume continue to decrease, but  $a$  is unchanged.

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