# On the existence of a Na -deficient monoclinic trinepheline with composition $\mathrm{Na}_{7.85} \mathrm{Al}_{7.85} \mathrm{Si}_{8.15} \mathbf{O}_{32}$ 

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#### Abstract

The crystal structure of a new synthetic $\mathrm{Na}_{8-r} \mathrm{Al}_{8-r} \mathrm{Si}_{8+r} \mathrm{O}_{32}(r=0.15)$ modification has been solved by direct methods and refined to an $R$ index of 0.0289 for 11643 independent reflections and 764 parameters. The compound is monoclinic with space group symmetry $P 112_{1}, a=9.9897(6) \AA, b=$ 9.9622(6) $\AA, c=24.979(2) \AA$, and $\gamma=119.788(4)^{\circ}$. The twinning of three individuals related by rotation around the $\mathbf{c}$ direction for $120^{\circ}$ was accounted for in the calculation. The phase was named monoclinic trinepheline because the length of its $c$ lattice parameter is $3 \times$ the length of the $c$ parameter in nepheline, whereas the $a$ parameter is almost the same in both structures, similarly to a previously investigated trinepheline of hexagonal symmetry. Actually, the present compound represents an intermediate state between these two phases. The characteristic of the crystal structure are layers of six-membered rings built up of regularly alternating $\mathrm{AlO}_{4}$ and $\mathrm{SiO}_{4}$ tetrahedra. The layers are perpendicular to [001] and are connected through apical O atoms. The resulting spatial network contains channels in which the Na cations are situated. The structure of monoclinic trinepheline can be described as a consecutive stacking of basic nepheline like modules (containing two tetrahedral sheets) and one sheet corresponding to those observed in the hexagonal trinepheline structure. The topology of the new compound corresponds to that of tridymite. The coordination sequences are identical for all T atoms in the asymmetric unit: 4-12-25-44-67-96-130-170-214-264. The vertex symbols for the tetrahedral centers are $6_{2} \cdot 6_{2} \cdot 6_{2} \cdot 6_{2} \cdot 6_{2} \cdot 6_{2}$. The framework density of the structure is equal to 22.2 T atoms $/ 1000 \AA^{3}$.


Keywords: Monoclinic trinepheline, $\mathrm{Na}_{7.85} \mathrm{Al}_{7.85} \mathrm{Si}_{8.15} \mathrm{O}_{32}$, flux method, nepheline

## INTRODUCTION

The crystal structure of natural nepheline of idealized composition $\mathrm{Na}_{6} \mathrm{~K}_{2} \mathrm{Al}_{8} \mathrm{Si}_{8} \mathrm{O}_{32}$ was reported by Hahn and Buerger (1955) in space group $P 6_{3}$. It can be described as a stuffed derivative of high-tridymite where half of the silicon ions are replaced by aluminum. Charge compensation is achieved by incorporation of alkali cations in the cavities of the structure- $75 \%$ of the cavities are populated by smaller Na and the remaining $25 \%$ by larger potassium cations. No pure Na nephelines are observed in the nature. Synthetically obtained Na-rich nephelines with the same basic structure have been studied by several authors (Dollase and Thomas 1978; Gregorkiewitz 1984; Roth 1985; Hippler and Böhm 1989; Dimitrijević et al. 2004). The investigations were mainly conducted because of nepheline's property to show Nasolid state ionic conductivity. The structural difference between natural samples and synthetic Na nephelines is based on the fact that excess Na in samples containing more than 6 Na apfu (see above) must be accommodated on the large alkali site. Sodium substitution involves either collapse of the hexagonal channels, or an off-centering of Na toward the channel wall (about $0.35 \AA$ ). This is due to the fact that ions smaller than potassium cannot establish a regular coordination with the surrounding O atoms, but form bond distances of noticeable different lengths.

[^0]According to Henderson and Roux (1977) and Henderson and Thompson (1980) pure $\mathrm{NaAlSiO}_{4}$ shows complex displacive polymorphic transitions, depending on the method of synthesis. Two different structures can exist at ambient conditions: an orthorhombic superstructure with $a \approx a_{0}, b \approx a_{0} \sqrt{3}$, and $c \approx 3 c_{0}$, ( $a_{0}$ and $c_{0}$ refer to the normal nepheline unit cell) and a second phase with even lower symmetry. The room-temperature phases transform to hexagonal symmetry at about 160 and $200^{\circ} \mathrm{C}$, respectively (Henderson and Roux 1977). The authors also mention a discontinuity in the thermal expansion near $870{ }^{\circ} \mathrm{C}$ and attribute it to a transformation into a high-tridymite-like structure. Dimitrijević et al. (2004) established the presence of a low-carnegieite phase below $850^{\circ} \mathrm{C}$, although nephelines in this work were synthesized by a different method based on thermal treatment of zeolitic material (LTA, FAU, and GIS types). Schneider et al. (1994) reported that single crystals of nepheline $\left(\mathrm{NaAlSiO}_{4}\right)$ are stable up to $1300{ }^{\circ} \mathrm{C}$, when they transform to carnegieite.

The orthorhombic structure of Henderson and Roux may be similar to a hydrothermally synthesized phase reported by Klaska (1974). Orthorhombic nepheline crystallizes in space group Pna2 ${ }_{1}$, and, according to Klaska, the structure is composed exclusively of tetrahedral rings of elliptical form, although no atomic coordinates were given. The structure of hexagonal trinepheline (Kahlenberg and Böhm 1998) is also built up of elliptical rings only. $\mathrm{NaAlSiO}_{4}$ forms showing hexagonal symmetry
with a tripling of the $\mathbf{c}$ axis were also reported by Jarchow et al. (1966) and Brown et al. (1972). The latter phase has a hexagonal cell with $a=b \approx a_{0} \sqrt{ } 3, c=3 c_{0}$. Selker et al. (1985) reported two monoclinic trinepheline compounds. The first structure is reported to crystallize in space group $P 2_{1}$ with pseudo hexagonal metric symmetry ( $a=9.990 \AA, b=25.100 \AA, c=9.990 \AA$, and $\beta=120.0^{\circ}$ ), while the other structure adopts space group $P n$ with pseudo orthorhombic symmetry ( $a=14.991 \AA, b=8.625$ $\AA, c=25.110 \AA$, and $\beta=90.2^{\circ}$ ). Selker (1987) gives detailed graphical description of the both structures without any atomic coordinates. The first structure is composed of ditrigonal and elliptical tetrahedral rings, while the other is composed of exclusively elliptical rings.

The goal of the present investigation is to clarify the structural relationships of the novel synthesized monoclinic trinepheline to already known nepheline-type phases.

## EXPERIMENTAL METHODS

As suggested by Roth (1985), the flux technique, using cryolite as hightemperature solvent, was applied for growth of the crystals. The starting materials were $\mathrm{Na}_{2} \mathrm{CO}_{3}$ (Merck, $99.95 \%$ ), $\mathrm{Al}_{2} \mathrm{O}_{3}$ (Fluka, 99.99\%), $\mathrm{SiO}_{2}$ (Alfa Aesar, 99.995\%) and natural cryolite $\mathrm{Na}_{3} \mathrm{AlF}_{6}$, (obtained from the Ivigtût area of Greenland). The reagents were carefully mixed in an agate mortar. Cryolite was added and the compounds were mixed again. A sample of 1 g with a nutrient-to-flux ratio of 2:5 was placed in a 50 mL platinum crucible with a platinum lid. The mixture was heated from 1073 to 1423 K over a period of 3.5 h in a resistance-heated furnace; it was held at this temperature for 4 h to homogenize the melt, then cooled down to 1323 K at a rate of $6 \mathrm{Kh}^{-1}$ for nepheline to crystallize. Subsequently, it was cooled down to 1073 K by a higher rate of $100 \mathrm{Kh}^{-1}$ and quenched immediately. The only obtained crystal was of pseudohexagonal platy habit ( $\approx 1 \mathrm{~mm}$ in diameter). It was mechanically removed from the polycrystalline cryolite matrix.

The composition of a crystal fragment was determined with a JEOL JXA 8100 electron microprobe using analytical conditions of 15 kV acceleration voltage and 5 nA beam current. Natural jadeite, natural quartz, and synthetic corundum were chosen as standards with measurement times of 20 s on peaks and 10 s on the backgrounds of the X-ray lines. To minimize beam damage of the sample surface, a rastered electron beam with a size adjusted to the size of the crystal fragment was used for the analyses. Raw counts were corrected using the PRZ correction procedure. X-ray images of the distribution of $\mathrm{Na}, \mathrm{Al}$, and Si in the investigated crystal fragment did not show significant spatial changes in composition. The average chemical composition based on six analyzed spots is $\mathrm{Na}_{7.85(1)} \mathrm{Al}_{7.85(1)} \mathrm{Si}_{8.15(1)} \mathrm{O}_{32}$ (Table 1).

Another fragment of the pseudohexagonal plate with good optical quality and irregular shape was selected for the structural investigations. X-ray diffraction data collection was performed with a two-circle Stoe IPDS-2 imaging-plate diffractometer at 302 K operated at 50 kV and 40 mA and with crystal-to-detector distance of 100 mm . A full sphere of three-dimensional data were collected using graphite monochromated $\mathrm{Mo} K \alpha$ radiation in two consecutive runs ( $\phi=0$ and $90^{\circ}$, respectively) in $\omega$ range from 0 to $180^{\circ}$ (frame width of $1^{\circ}$ and count-time of 5 min per frame). Intensity integration and data reduction was done using the Stoe X-Area (Stoe and Cie GmbH 2005) software. Even though material of the aforementioned chemical composition has a low linear absorption coefficient for Mo-radiation, intensities were corrected for absorption using an analytical numerical absorption correction based on 37 indexed faces precisely describing the morphology of the irregular fragment used for data collection. In the range $|h|$ $\leq 13 ;|k| \leq 13 ;|l| \leq 34,39733$ reflections were measured, 36564 of these had $I>$ $4 \sigma(I)$. Crystal lattice constants were showing hexagonal metric symmetry, but the highest Laue symmetry with an acceptable $R_{\text {int }}$ value of 0.032 corresponds to $112 / \mathrm{m}$,
while the $R_{\text {int }}$ value for Laue symmetry $6 / m$ was 0.418 . The systematic absences ( $00 l$ ), $l \neq 2 n$ resulted in two possible space groups: $P 112_{1}$ or $P 112_{1} / m$ (orientation corresponding to c unique).

## STRUCTURE SOLUTION, REFINEMENT, AND TWINNING

Determination of a model for the whole structure was done by direct methods, using the program SIR2004 (Burla et al. 2005). E-statistics did not clearly indicate the presence of a center of symmetry. Therefore, structure solution was started in both possible space groups. The best solution in space group $P 112_{1}$ (with $R$-value of 0.093 ) resulted in an $E$-map of which the most intense peaks were describing a stuffed framework structure. The further refinement of the model was carried out with the SHELXL97 program (Sheldrick 1998). X-ray scattering factors for the different cations, and their anomalous dispersion corrections were taken from the International Tables for Crystallography (Prince 2004); the values for $\mathrm{O}^{2-}$ were taken from Hovestreydt (1983). The starting model seemed to be reasonable, but iterative full-matrix least-squares calculation based on $F^{2}$ using isotropic displacement factors converged to an unconvincing unweighted $R_{1}$ index of about 0.10 . Introduction of anisotropic displacement parameters for all atoms resulted in a slight decrease of $R_{1}(\approx 0.08)$. However, almost one fifth of the framework ions had non positive definite values for the principal mean square atomic displacement parameters.

To explain the difficulties during the refinement, a possible twinning was considered. The unit cell is pseudohexagonal, with $a \approx b$ and $\gamma \approx 120^{\circ}$. Twinning of three individuals by rotation around the threefold axis coinciding with $\mathbf{c}$ is very common in such cases. After the introduction of the corresponding twin law (using TWIN and BASF commands implemented in SHELX97), the residuals $R_{1}$ and $w R_{2}$ for the refinement based on $F^{2}$ using isotropic displacement for all the reflections dropped to 0.050 and 0.121 , respectively. The refined volume fractions for the twin components were $87.9,7.0$, and $5.1 \%$, respectively.

Following the results of the chemical analysis (indicating a Na deficiency), site occupation numbers of three Na atoms were refined. However, a constraint for the total Na content according to the composition derived from the EMP measurement was used. These three sites correspond to the positions of potassium in natural nepheline, and according to Hippler and Böhm (1989) in the case of Na deficiency vacancies are observed on these positions. Indeed, the refinements indicated a depletion of Na on two of these sites.

The final refinement using anisotropic displacement parameters converged at $R_{1}=0.0289$ and $w R_{2}=0.0694$ for all 11643 independent reflections and 764 refined parameters. Experimental details are summarized in Table 2. The refined atomic coordinates, site occupancies, equivalent isotropic and anisotropic displacement parameters, as well as selected bond lengths and angles are given in Tables 3-6.

## DESCRIPTION AND DISCUSSION OF THE STRUCTURE

The crystal structure of monoclinic trinepheline is built up by a sequence of tetrahedral layers perpendicular to the c direction. Each layer is composed of different types of sixfold rings made of $\mathrm{SiO}_{4}$ and $\mathrm{AlO}_{4}$ tetrahedra. The sequence of directedness of the tetrahedra is $U D U D U D$. Among the six layers corresponding to the translation period along [001] two blocks of 3 sheets can be distinguished, which are related by a $2_{1}$ screw axis. Two layers of each block are composed of two different types of sixmembered tetrahedral rings. Of the four crystallographically independent rings comprising these two layers one exhibits a slightly distorted ditrigonal symmetry, while the other three rings are elliptical, and related to each other through a pseudothreefold axis. This feature results in almost trigonal local layer symmetry. The third sheet situated between the two "trigonal" layers is exclusively composed of elliptical rings. Within this

Table 1. Results of the electron microprobe analyses (given as wt\% oxides) of the monoclinic trinepheline flux-grown crystal

| Oxide | Analysis 1 | Analysis 2 | Analysis 3 | Analysis 4 | Analysis 5 | Analysis 6 | Calculated |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{Na}_{2} \mathrm{O}$ | 21.43 | 21.38 | 21.54 | 21.47 | 21.35 | 21.47 |  |
| $\mathrm{SiO}_{2}$ | 43.14 | 43.01 | 43.54 | 43.01 | 42.92 | 43.41 |  |
| $\mathrm{Al}_{2} \mathrm{O}_{3}$ | 35.32 | 35.08 | 35.45 | 35.25 | 35.09 | 35.43 |  |
| $\sum_{\text {Formula }}$ | $\mathrm{Na}_{7.85(1)} \mathrm{Al}_{7.85(1)} \mathrm{Si}_{8.15(1)} \mathrm{O}_{32}$ | 99.47 |  | 100.53 | 99.73 | 99.36 | 100.40 |

TABLE 2. Experimental details

| Crystal data |  |
| :---: | :---: |
| Chemical formula | $\mathrm{Na}_{7.85} \mathrm{Al}_{7.85} \mathrm{Si}_{8.15} \mathrm{O}_{32}$ |
| $M_{r}$ | 1133.18 |
| Cell setting, space group | Monoclinic, $P 112_{1}, C$ unique (no. 4) |
| $a, b, c(\AA)$ | 9.9897(6), 9.9622(6), 24.979(2) |
| $\gamma\left({ }^{\circ}\right)$ | 119.788(4) |
| $V\left(\AA^{3}\right)$ | 2157.4(2) |
| $Z$ | 3 |
| $D_{x}\left(\mathrm{Mg} / \mathrm{m}^{3}\right)$ | 2.617 |
| Radiation type | MoK $\alpha$ |
| No. of reflections for cell parameters | 14178 |
| $\theta$ range ( ${ }^{\circ}$ ) | 1.63-29.19 |
| $\mu\left(\mathrm{mm}^{-1}\right)$ | 0.873 |
| Temperature (K) | 302(2) |
| Crystal form, color | irregular, white |
| Crystal size (mm) | $0.23 \times 0.2 \times 0.17$ |
| Data collection |  |
| Diffractometer | Stoe IPDS-2 |
| Data collection method | Rotation method |
| Absorption correction | Integration |
| $T_{\text {min }}$ | 0.871 |
| $T_{\text {max }}$ | 0.917 |
| No. of measured and | 39733, 11643, 11519 |
| independent reflections | reflections with $/>4 \sigma(I)$ |
| $R_{\text {int }}$ | 0.032 |
| $\theta_{\text {max }}\left({ }^{\circ}\right)$ | 29.2 |
| Range of $h, k, l$ | $-13 \leq h \leq 13$ |
|  | $-13 \leq k \leq 13$ |
|  | $-34 \leq 1 \leq 34$ |
| Refinement |  |
| Refinement on | $F^{2}$ |
| $R_{1}$ (all), $R_{1}\left[F^{2}>4 \sigma\left(F^{2}\right)\right]$ | 0.029, 0.028 |
| $w R_{2}(\mathrm{all}), w R_{2}\left[F^{2}>4 \sigma\left(F^{2}\right)\right]$ | 0.069, 0.069 |
| $S$ | 1.084 |
| No. of reflections | 11643 |
| No. of parameters | 764 |
| Weighting scheme | $\begin{aligned} & W=1 /\left[\sigma^{2}\left(F_{o}^{2}\right)+(0.0305 \mathrm{P})^{2}+1.9644 \mathrm{P}\right] ; \\ & \mathrm{P}=\left[2 F_{c}^{2}+F_{o}^{2}\right] / 3 \end{aligned}$ |
| $(\Delta / \sigma)_{\text {max }}$ | 0.089 |
| $\Delta \rho_{\max ,} \Delta \rho_{\min }\left(\mathrm{e} \AA^{-3}\right)$ | 0.981, -0.835 |

layer two different orientations of the elliptical rings can be distinguished. The longest axes of the rings may be parallel or perpendicular to each other. Figure 1 shows projections of the three basic tetrahedral sheets of one block (using the computer program STRUPLO91, Fischer et al. 1991). For each of the three layers simplified skeletal representations are given to make the relation of the six-membered rings more obvious.

Connection of the layers is achieved through apical O atoms. The angles between the horizontal (001) and the planes defined by the three basal O atoms of the tetrahedra vary from 5 to $17^{\circ}$. The layers are interconnected by the apical atoms O4a-c, O5a-c, O9a-c, and O14a-c and their symmetrical equivalents. One half of the paired tetrahedra belonging to different adjacent layers can be classified as a cis arrangement, whereas the remaining half achieves a trans configuration (Flörke 1967). From the T-O bonds in Table 5 it can be seen that two groups of twelve tetrahedra can be distinguished. The first group has mean values for T-O bond lengths of about 1.62, while the second averages of about $1.74 \AA$. The average Al content for the two groups was calculated to $0.09(3)$ and $0.84(3)$, respectively (Jones 1968). This arrangement corresponds to an excess in Si , which is in agreement with the chemical analysis. Al-rich tetrahedra are linked with four Si-rich tetrahedra and vice versa, obeying the aluminum avoidance rule (Loewenstein 1954). This results in a


Figure 1. The basic three building layers of monoclinic trinepheline. Hatched and white tetrahedra correspond to $\mathrm{SiO}_{4}$ and $\mathrm{AlO}_{4}$, respectively. On the right side, skeletal representations are given to outline the positions of tetrahedrally coordinated cations: (a) $0.00 \leq$ $z \leq 0.24$; (b) $0.16 \leq z \leq 0.40$; (c) $0.32 \leq z \leq 0.56$.
polar crystal with all Al- and Si-tetrahedra pointing in different directions along the $\mathbf{c}$ axis.

The average values of the O-T-O angles for all 24 tetrahedra are close to the ideal value of $109.47^{\circ}$, but the individual O-T-O angles range from 106.0 to $114.0^{\circ}$ for the $\mathrm{SiO}_{4}$ groups and from 103.5 to $116.8^{\circ}$ for the $\mathrm{AlO}_{4}$ tetrahedra (Table 5). The polyhedral distortion is more obvious for tetrahedra containing Al. The mean tetrahedral angle variance (Robinson et al. 1971) for the $\mathrm{SiO}_{4}$ and $\mathrm{AlO}_{4}$ tetrahedra has values of 2.95 and 8.41 , respectively. The T-O-T angles vary between 128.9 and $157.0^{\circ}$ (average: $139.7^{\circ}$ ). This value is only slightly lower than the value of $142^{\circ}$ suggested by Liebau (1985) for "unstrained" Si-O-Al angles.

The framework density of the present porous crystal structure is 22.2 T atoms $/ 1000 \AA^{3}$. The connectivity of the T atoms within the framework can be characterized by the so-called coordination sequences, which are identical for all $\mathrm{Al} / \mathrm{Si}$-atoms in the asymmetric unit: 4-12-25-44-67-96-130-170-214-264. The vertex symbols for the tetrahedral centers are $6_{2} \cdot \sigma_{2} \cdot \sigma_{2} \cdot 6_{2} \cdot 6_{2} \cdot 6_{2}$. In summary, monoclinic trinepheline can be classified as a threedimensional four-connected net with tridymite topology.

Charge balance in the structure of trinepheline is achieved by incorporation of Na ions located in the channels running parallel c [Fig. 2, prepared using the computer program DIAMOND (Crystal Impact 2005)]. Within these tunnels every Na is hosted between two six-membered tetrahedral rings. Regardless of the elliptical or ditrigonal distortion of the rings, Na always occupies an off-centered position, because the dimensions of the cavities are too big for it to take the central position. Therefore, Na atoms are irregularly coordinated. Among them two groups could be

TABLE 3. Fractional coordinates $\left(\times 10^{4}\right)$, site occupancies, and equivalent isotropic displacement parameters ( $\AA^{2} \times 10^{3}$ )

| Atom | $x$ | $y$ | $z$ | Occ | $U_{\text {eq }}$ | Atom | $x$ | $y$ | $z$ | Occ | $U_{\text {eq }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Si1a | 7561(2) | 6722(2) | 8687(1) | 1.000 | 10(1) | O3a | 9333(4) | 7126(4) | 8648(2) | 1.000 | 17(1) |
| Si1b | 6497(2) | 8996(2) | 344(1) | 1.000 | 11(1) | O3b | 7122(4) | 7785(4) | 315(2) | 1.000 | 20(1) |
| Si1c | 981(1) | 8031(1) | 7077(1) | 1.000 | 10(1) | O3c | 2431(4) | 9746(3) | 7039(2) | 1.000 | 15(1) |
| Si2a | 6691(1) | 3381(1) | 377(1) | 1.000 | 11(1) | O4a | 6978(4) | 6869(4) | 8096(1) | 1.000 | 16(1) |
| Si2b | 7044(1) | 3494(1) | 7006(1) | 1.000 | 11(1) | 04b | 6666(2) | 9723(2) | 9750(2) | 1.000 | 19(1) |
| Si2c | 3173(2) | 6453(2) | 8739(1) | 1.000 | 13(1) | O4c | 408(4) | 7450(4) | 6471(1) | 1.000 | 18(1) |
| Si3a | 836(1) | 7548(1) | 342(1) | 1.000 | 11(1) | O5a | 9613(4) | 2582(4) | 8025(2) | 1.000 | 18(1) |
| Si3b | 2349(1) | 3209(1) | 6990(1) | 1.000 | 10(1) | 05b | 2391(2) | 2594(2) | 9741(2) | 1.000 | 21(1) |
| Si3c | 3337(2) | 967(1) | 8691(1) | 1.000 | 10(1) | O5c | 6541(5) | 9480(4) | 6406(1) | 1.000 | 19(1) |
| Si4a | 8853(1) | 2266(1) | 8617(1) | 1.000 | 10(1) | O6a | 7409(5) | 7882(4) | 9107(2) | 1.000 | 22(1) |
| Si4b | 7915(1) | 6932(1) | 5333(1) | 1.000 | 10(1) | 06b | 2511(4) | 9696(4) | 5780(1) | 1.000 | 19(1) |
| Si4c | 6318(1) | 8752(1) | 7000(1) | 1.000 | 10(1) | O6c | 9604(3) | 8118(3) | 7393(2) | 1.000 | 13(1) |
| Al1a | 6539(2) | 9027(2) | 9106(1) | 1.000 | 10(1) | 07a | 3305(4) | 5029(4) | 9002(2) | 1.000 | 19(1) |
| Al1b | 1137(2) | 7767(2) | 5822(1) | 1.000 | 11(1) | 07b | 5086(4) | 1771(4) | 441(2) | 1.000 | 20(1) |
| Al1c | 7631(2) | 6809(2) | 7456(1) | 1.000 | 10(1) | 07c | 8535(4) | 3275(4) | 7062(2) | 1.000 | 20(1) |
| Al2a | 846(2) | 7536(2) | 9080(1) | 1.000 | 10(1) | 08a | 7786(4) | 673(4) | 5367(2) | 1.000 | 16(1) |
| Al2b | 2458(2) | 3301(2) | 5749(1) | 1.000 | 11(1) | 08b | 515(4) | 1949(4) | 7026(2) | 1.000 | 18(1) |
| Al2c | 3704(2) | 1290(1) | 7440(1) | 1.000 | 10(1) | O8c | 2563(5) | 2063(4) | 8644(2) | 1.000 | 19(1) |
| Al3a | 2082(2) | 3054(2) | 9106(1) | 1.000 | 12(1) | 09a | 172(2) | 7179(2) | 9739(2) | 1.000 | 23(1) |
| Al3b | 6702(2) | 9054(2) | 5741(1) | 1.000 | 10(1) | 09b | 2993(4) | 3136(4) | 6399(1) | 1.000 | 16(1) |
| Al3c | 9000(2) | 1992(1) | 7366(1) | 1.000 | 10(1) | O9c | 3568(5) | 564(5) | 8086(2) | 1.000 | 23(1) |
| Al4a | 6708(2) | 3424(2) | 9092(1) | 1.000 | 11(1) | 010a | 8565(4) | 3644(4) | 5501(2) | 1.000 | 25(1) |
| Al4b | 6874(2) | 3575(2) | 5707(1) | 1.000 | 11(1) | O10b | 2581(4) | 4912(4) | 7077(2) | 1.000 | 22(1) |
| Al4c | 2970(2) | 6550(2) | 7435(1) | 1.000 | 12(1) | O10c | 4973(4) | 1766(4) | 8995(2) | 1.000 | 19(1) |
| Na1a | 9921(1) | 4366(1) | 9717(1) | 1.000 | 23(1) | O11a | 8204(4) | 3437(4) | 8732(2) | 1.000 | 19(1) |
| Na1b | 5617(3) | 5500(3) | 6300(1) | 1.000 | 23(1) | 011b | 6726(4) | 5094(4) | 5425(2) | 1.000 | 18(1) |
| Na1c | 5459(3) | 9763(3) | 8033(1) | 1.000 | 26(1) | O11c | 4593(4) | 8218(4) | 7204(2) | 1.000 | 20(1) |
| Na 2 a | 5302(1) | 5497(1) | 9730(1) | 1.000 | 24(1) | O12a | 4626(4) | 8076(5) | 8920(2) | 1.000 | 28(1) |
| Na 2 b | 4511(3) | 147(3) | 6410(1) | 1.000 | 25(1) | O12b | 8060(4) | 3423(4) | 743(2) | 1.000 | 17(1) |
| Na 2 c | 9594(3) | 5273(3) | 8031(1) | 1.000 | 26(1) | O12c | 7285(4) | 4934(4) | 7375(2) | 1.000 | 22(1) |
| Na3a | 4411(3) | 4527(2) | 8149(1) | 1.000 | 22(1) | 013a | 7442(4) | 508(4) | 8650(2) | 1.000 | 18(1) |
| Na3b | 4298(1) | 9931(1) | 9722(1) | 1.000 | 21(1) | 013b | 9654(4) | 7277(4) | 5363(2) | 1.000 | 19(1) |
| Na 3 c | 353(3) | 4702(3) | 6401(1) | 1.000 | 27(1) | O13c | 6646(4) | 7301(4) | 6990(1) | 1.000 | 13(1) |
| Na4a | 9637(3) | 9596(3) | 9723(2) | 0.852(5) | 57(1) | 014a | 7232(2) | 3582(2) | 9759(2) | 1.000 | 22(1) |
| Na4b | 9828(6) | 9842(5) | 6433(2) | 1.000 | 91(2) | 014b | 6861(5) | 3854(5) | 6399(2) | 1.000 | 29(1) |
| Na4c | 431(4) | 355(4) | 7974(2) | 0.923(5) | 54(1) | O14c | 3231(6) | 6273(6) | 8106(2) | 1.000 | 34(1) |
| O1a | 6450(4) | 4956(4) | 8904(2) | 1.000 | 20(1) | 015a | 144(4) | 2527(4) | 9060(1) | 1.000 | 16(1) |
| 01b | 4706(4) | 8168(4) | 524(2) | 1.000 | 23(1) | 015b | 7580(4) | 7915(4) | 5780(1) | 1.000 | 16(1) |
| O1c | 1380(4) | 6819(4) | 7367(2) | 1.000 | 18(1) | O15c | 7510(4) | 75(3) | 7403(1) | 1.000 | 14(1) |
| O2a | 509(4) | 2701(5) | 5761(2) | 1.000 | 20(1) | O16a | 1571(4) | 6340(4) | 8917(2) | 1.000 | 24(1) |
| O2b | 3264(4) | 2780(4) | 7433(2) | 1.000 | 16(1) | 016b | 3588(4) | 5200(4) | 5557(2) | 1.000 | 21(1) |
| O2c | 2198(5) | 9464(4) | 9038(2) | 1.000 | 24(1) | O16c | 5536(4) | 1915(4) | 7193(2) | 1.000 | 20(1) |



Figure 2. Crystal structure of monoclinic trinepheline viewed parallel to c. Sodium ions are represented by thermal ellipsoids (50\% probability level).
distinguished. Na1a-Na3c, located in the less symmetrical channels, are surrounded by six or seven O atoms. The interatomic distances are ranging from 2.3 to $2.9 \AA$. The remaining Na atoms found in the channels created by ditrigonal tetrahedral rings have an even more irregular coordination. Four shorter bonds between 2.3 and 2.7 (close to the typical average bond lengths for $\mathrm{Na}-\mathrm{O}$
of 2.5-2.6 $\AA$ according to Shannon and Prewitt 1969) could be distinguished from additional three to four interatomic distances ranging from 2.8 to $3.1 \AA$. This unusual coordination of Na situated in the ditrigonal channels is forced by the geometry of these channels. Pluth and Smith (1980) reported similar coordination environments for Na in the case of dehydrated zeolite A. The very irregular coordination was due to statistical occupation of the positions.

Bond-valence sums (BVS) were calculated based on the parameter set of Brown (1981) using the program IVTON (Balić-Žunić and Vicković 1996). The results are given in the Table 7. For the Si designated sites the BVS values are very close to 4.0 v.u. Al-rich sites have BVS about 3.2 v.u., clearly indicating some excess of silicon in these positions, as it was expected from chemical analysis. The nine Na atoms situated in between elliptical rings only have bond-valence sums ranging from $0.8-1.0$ v.u., while for $\mathrm{Na} 4 a-\mathrm{c}$ the sums are about 0.5 v.u. A similar BVS value can be observed for Na4 atom in hexagonal trinepheline (Kahlenberg and Böhm 1998) situated in the pseudo hexagonal channels of the structure, as well as for Na2 in pure Na nephelines (Hippler and Böhm 1989) located in the ditrigonal channels. This points to a pronounced underbonding of $\mathrm{Na} 4 \mathrm{a}-\mathrm{c}$ in this structure.

The mean square displacements for thermal motions were

TABLE 4. Anisotropic displacement parameters $\left(\AA^{2} \times 10^{3}\right)$

| Atom | $U_{11}$ | $U_{22}$ | $U_{33}$ | $U_{23}$ | $U_{13}$ | $U_{12}$ | Atom | $U_{11}$ | $U_{22}$ | $U_{33}$ | $U_{23}$ | $U_{13}$ | $U_{12}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Sila | 14(1) | 12(1) | 9(1) | 0 (1) | 1(1) | 9(1) | O3a | 17(1) | 23(1) | 13(1) | -1(1) | -2(1) | 12(1) |
| Si1b | 13(1) | 9(1) | 10(1) | -1(1) | -1(1) | 5(1) | O3b | 32(2) | 23(1) | 13(1) | 4(1) | 2(1) | 21(1) |
| Si1c | 9(1) | 9(1) | 10(1) | $0(1)$ | -1(1) | 4(1) | O3c | 12(1) | 10(1) | 19(2) | 1(1) | -3(1) | 2(1) |
| Si2a | 10(1) | 9(1) | 13(1) | -1(1) | $-1(1)$ | 4(1) | O4a | 17(2) | 25(2) | 10(1) | -1(1) | -1(1) | 14(1) |
| Si2b | 10(1) | 10(1) | 14(1) | -1(1) | 0(1) | 5(1) | O4b | 34(1) | 20(1) | 12(1) | $-1(1)$ | $0(2)$ | 19(1) |
| Si2c | 11(1) | 11(1) | 18(1) | 1(1) | -1(1) | 7(1) | O4c | 15(1) | 20(2) | 10(1) | $0(1)$ | 1(1) | 1(1) |
| Si3a | 9(1) | 11(1) | 11(1) | $0(1)$ | -1(1) | 5(1) | O5a | 15(1) | 19(2) | 12(1) | -4(1) | O(1) | 1(1) |
| Si3b | 11(1) | 11(1) | 10(1) | $0(1)$ | 0(1) | 7(1) | 05b | 30(1) | 35(1) | 10(1) | -3(2) | -4(2) | 25(1) |
| Si3c | 13(1) | 9(1) | 9(1) | $0(1)$ | -1(1) | 6(1) | O5c | 33(2) | 24(2) | 8(1) | $0(1)$ | 4(1) | 20(1) |
| Si4a | 8(1) | 9(1) | 9(1) | $0(1)$ | $0(1)$ | 2(1) | O6a | 40(2) | 33(2) | 12(1) | -5(1) | -7(1) | 32(2) |
| Si4b | 12(1) | 12(1) | 8(1) | $0(1)$ | -1(1) | 7(1) | 06b | 22(2) | 14(1) | 13(1) | 1(1) | $0(1)$ | 2(1) |
| Si4c | 11(1) | 10(1) | 10(1) | $0(1)$ | -1(1) | 5(1) | O6c | 8(1) | 10(1) | 18(2) | -1(1) | $0(1)$ | 2(1) |
| Al1a | 12(1) | 9(1) | 10(1) | $0(1)$ | $0(1)$ | $6(1)$ | 07a | 8(1) | 14(1) | 33(2) | 7(1) | 2(1) | 4(1) |
| Al1b | 13(1) | 12(1) | 10(1) | $0(1)$ | 1(1) | 7(1) | 07b | 15(2) | 16(2) | 29(2) | -1(1) | 9(1) | 8(1) |
| Al1c | 11(1) | 11(1) | 9(1) | $0(1)$ | O(1) | 6(1) | 07c | 16(1) | 17(1) | 30(2) | 11(1) | 6(1) | 9(1) |
| Al2a | 10(1) | 9(1) | 10(1) | $0(1)$ | -1(1) | 4(1) | O8a | 19(1) | 10(1) | 16(1) | 2(1) | 2(1) | 4(1) |
| Al2b | 10(1) | 11(1) | 10(1) | $0(1)$ | 0(1) | 5(1) | 08b | 14(1) | 26(2) | 15(2) | -5(1) | 3(1) | 11(1) |
| Al2c | 11(1) | 12(1) | 8(1) | -1(1) | $0(1)$ | 6(1) | O8c | 29(2) | 22(2) | 11(1) | $-1(1)$ | $0(1)$ | 17(1) |
| Al3a | 14(1) | 14(1) | 10(1) | $0(1)$ | 2(1) | 8(1) | 09a | 17(1) | 24(1) | 12(1) | 2(2) | 4(2) | $-1(1)$ |
| $\mathrm{Al3b}$ | 10(1) | 10(1) | 10(1) | $0(1)$ | $0(1)$ | 4(1) | 09b | 14(1) | 22(2) | 11(1) | 1(1) | $0(1)$ | 9(1) |
| Al3c | 9(1) | 9(1) | 11(1) | 1(1) | 2(1) | 4(1) | O9c | 47(2) | 28(2) | 11(1) | $0(1)$ | -2(1) | 32(2) |
| Al4a | 9(1) | 11(1) | 11(1) | 1(1) | $0(1)$ | 5(1) | O10a | 13(1) | 17(2) | 44(2) | -1(2) | 3(1) | 7(1) |
| Al4b | 10(1) | 11(1) | 11(1) | -1(1) | -1(1) | 4(1) | O10b | 29(2) | 15(1) | 24(2) | -3(1) | 1(1) | 14(1) |
| Al4c | 13(1) | 12(1) | 12(1) | $-1(1)$ | -3(1) | 8(1) | O10c | 13(1) | 14(1) | 22(2) | 4(1) | $-1(1)$ | 1(1) |
| Na1a | 23(1) | 23(1) | 26(1) | $0(1)$ | $0(1)$ | 14(1) | O11a | 16(1) | 16(2) | 24(2) | 3(1) | 4(1) | 9(1) |
| Na1b | 26(1) | 24(1) | 19(1) | -7(1) | -3(1) | 11(1) | O11b | 23(2) | 14(1) | 19(2) | -2(1) | -4(1) | 11(1) |
| Na1c | 22(1) | 38(1) | 18(1) | -7(1) | -4(1) | 16(1) | O11c | 10(1) | 12(1) | 34(2) | -3(1) | 1(1) | 3(1) |
| Na 2 a | 24(1) | 28(1) | 19(1) | -3(1) | $-1(1)$ | 12(1) | O12a | 11(2) | 18(2) | 51(3) | -4(2) | -1(2) | 5(1) |
| Na 2 b | 22(1) | 40(1) | 20(1) | -2(1) | -2(1) | 20(1) | O12b | 18(1) | 16(2) | 22(2) | -5(1) | -6(1) | 12(1) |
| Na2c | 31(1) | 24(1) | 24(1) | 12(1) | 9(1) | 15(1) | O12c | 23(2) | 11(1) | 28(2) | $-1(1)$ | -2(1) | 6(1) |
| Na3a | 21(1) | 14(1) | 24(1) | -2(1) | 2(1) | 4(1) | O13a | 22(2) | 10(1) | 14(1) | $0(1)$ | 1(1) | 3(1) |
| Na3b | 21(1) | 27(1) | 17(1) | -1(1) | -2(1) | 13(1) | O13b | 17(1) | 26(2) | 16(1) | -3(1) | -2(1) | 11(1) |
| Na 3 c | 29(1) | 23(1) | 25(1) | 3(1) | 9(1) | 10(1) | O13c | 16(1) | 16(1) | 9(1) | -2(1) | -5(1) | 9(1) |
| Na 4 a | 66(1) | 67(1) | 37(1) | -2(2) | 0(3) | 32(1) | O14a | 21(1) | 23(1) | 13(1) | O(1) | -1(1) | 5(1) |
| Na4b | 174(4) | 95(3) | 62(2) | -51(2) | -53(2) | 111(3) | O14b | 40(2) | 39(2) | 16(2) | -1(1) | -5(1) | 25(2) |
| Na 4 c | 50(1) | 74(2) | 54(2) | -36(2) | -30(1) | 43(1) | O14c | 46(2) | 63(3) | 6(1) | -2(1) | -6(1) | 36(2) |
| O1a | 18(2) | 17(1) | 29(2) | 7(1) | 5(1) | 12(1) | O15a | 9(1) | 24(2) | 12(1) | 1(1) | -2(1) | 6(1) |
| O1b | 17(2) | 14(2) | 33(2) | 4(1) | 8(2) | 5(1) | O15b | 25(2) | 21(2) | 11(1) | $-1(1)$ | -2(1) | 19(1) |
| O1c | 8(1) | 15(1) | 32(2) | 3(1) | 1(1) | 5(1) | O15c | 16(1) | 10(1) | 11(1) | $0(1)$ | 1(1) | 3(1) |
| O2a | 10(1) | 28(2) | 19(2) | 5(1) | $0(1)$ | 7(1) | O16a | 13(2) | 12(2) | 47(2) | -2(1) | 5(1) | 6(1) |
| O2b | 22(2) | 20(1) | 13(1) | -3(1) | -3(1) | 15(1) | O16b | 13(2) | 16(1) | 32(2) | 5(1) | 4(1) | 8(1) |
| O 2 c | 19(2) | 16(1) | 26(2) | $0(1)$ | $0(1)$ | 1(1) | O16c | 11(1) | 11(1) | 36(2) | -3(1) | 5(1) | 5(1) |

calculated from the anisotropic displacement parameters (Table 4). Most of oxygen thermal motion ellipsoids are prolate with the longest axes perpendicular to the T-O direction and the shortest axes nearly parallel to the bonds. It is reasonable to assume that tetrahedra execute a librational motion about the T ion center (Busing and Levy 1964).

The apical O atoms O14b and O14c show a deviation from this general rule. Their anisotropic displacement parameters are considerably higher than the rest of the O atoms. They connect "trigonal" and "elliptical" layers and correspond to the statistical triple split position O1 in the model of Hahn and Buerger (1955) for natural nepheline. O14a connects two "trigonal" layers and shows no deviation with regard to the displacement parameters of the other oxygen atoms in the structure.

The strongly underbonded Na ions $\mathrm{Na} 4 \mathrm{a}-\mathrm{c}$ also have the highest values for the atomic displacements observed, as is often observed for guest ions in tunnel-like host structures. The lengths of principal axis for these atoms range from 0.15 to $0.46 \AA$ and lead to the conclusion that $\mathrm{Na} 4 \mathrm{a}-\mathrm{c}$ sodium ions are disordered. The thermal ellipsoid for Na 4 b exhibits the most pronounced elongation. This Na is located $0.18 \AA$ away from the center of the channel, unlike the other two ions that are off-centered for 0.38 $\AA$. Assuming that these positions represent an average in space and time, refinement with two split ions with isotropic thermal
parameters was implemented. The results indicated two separate, but also differently populated positions. However, the obtained $R$-values were clearly higher, so we concluded that at this level of investigation it is not possible to make a final conclusion about the nature of disorder of the Na ions.

The crystal structure of monoclinic trinepheline (Fig. 1) can be described as a simple stacking of the basic Hahn and Buerger model for natural nepheline composed of two layers (Fig. 3) and one layer (Fig. 4) corresponding to those observed in the hexagonal trinepheline model (Kahlenberg and Böhm 1998). This sequence is further transformed by the means of the $2_{1}$ screw axis, to form an additional three layers, giving a complete set of six layers fully defining the structure of monoclinic trinepheline. With this respect it is interesting to note that in hexagonal trinepheline a six-layer stacking sequence is realized preserving the hexagonal symmetry although the layers exclusively contain elliptical rings. The present structure can be understood as a "frozen" between classical nepheline and hexagonal trinepheline. It should be underlined that monoclinic trinepheline is in that sense closer to natural nepheline.

It is also important to note that the structure described in this paper is very similar to monoclinic trinepheline reported by Selker et al. (1985) and later by Selker (1987). However, as already stated in the introduction, Selker (1987) gave only a

Table 5. Selected bond lengths ( $\AA$ )

| Si1a-O3a | 1.611(4) | Si1b-O6b ${ }^{\text {i }}$ | 1.607(4) | Si1c-O3c | 1.603(3) |
| :---: | :---: | :---: | :---: | :---: | :---: |
| Si1a-04a | 1.620(4) | Si1b-O3b | 1.612(4) | Si1c-01c | 1.619(4) |
| Si1a-O6a | 1.622(4) | Si1b-01b | 1.618(4) | Si1c-O4c | 1.620(4) |
| Si1a-01a | 1.637(4) | Si1b-O4bi | 1.623(5) | Si1c-O6C ${ }^{\text {iii }}$ | 1.625(3) |
| Mean | 1.623 | Mean | 1.615 | Mean | 1.617 |
| Si2a-O14a ${ }^{\text {ii }}$ | 1.616(5) | Si2b-014b | 1.591(4) | Si2c-014c | 1.596(4) |
| Si2a-07b | 1.617(4) | Si2b-016c | 1.614(4) | Si2c-012a | 1.610(4) |
| Si2a-O12b | 1.628(4) | Si2b-07c | 1.616(4) | Si2c-016a | 1.611(4) |
| Si2a-O16biv | 1.634(4) | Si2b-O12c | 1.618(4) | Si2c-07a | 1.628(4) |
| Mean | 1.624 | Mean | 1.610 | Mean | 1.611 |
| Si3a-09aii | 1.613(5) | Si3b-010b | 1.608(3) | Si3c-O10c | 1.608(4) |
| Si3a-08a ${ }^{\text {i }}$ | 1.617(4) | Si3b-02b | 1.624(4) | Si3c-09c | 1.609(4) |
| Si3a-02av | 1.622(4) | Si3b-09b | $1.626(4)$ | $\mathrm{Si3c}-02 \mathrm{c}^{\text {vi }}$ | 1.610(4) |
| Si3a-O10a ${ }^{\text {iv }}$ | 1.623(4) | Si3b-08b | 1.630(3) | Si3c-08c | 1.626(4) |
| Mean | 1.619 | Mean | 1.622 | Mean | 1.614 |
| Si4a-O13a | 1.615(4) | Si4b-013b | 1.597(4) | Si4c-011c | 1.612(4) |
| Si4a-011a | 1.616(4) | Si4b-015b | 1.626(4) | Si4c-O15c ${ }^{\text {viii }}$ | 1.616(4) |
| Si4a-015a ${ }^{\text {vii }}$ | 1.620(4) | Si4b-O5biv | $1.626(5)$ | Si4c-O5c | 1.618(4) |
| Si4a-05a | 1.621(4) | Si4b-011b | 1.630(4) | Si4c-013c | 1.633(3) |
| Mean | 1.618 | Mean | 1.620 | Mean | 1.620 |
| Al1a-O12a | 1.722(4) | Al1b-O6b | 1.722(4) | Al1c-O12c | 1.735(4) |
| Al1a-O13a ${ }^{\text {viii }}$ | 1.722(4) | Al1b-O4c | 1.739(4) | Al1c-04a | 1.739(4) |
| Al1a-O4b | $1.732(5)$ | Al1b-O13b ${ }^{\text {iii }}$ | 1.741(4) | Al1c-O13c | 1.744(4) |
| Al1a-06a | 1.745(4) | Al1b-O12bix | 1.741(4) | Al1c-O6c | 1.749(3) |
| Mean | 1.730 | Mean | 1.736 | Mean | 1.741 |
| Al2a-O2c | 1.716(4) | Al2b-O16b | 1.721(4) | Al2c-016c | 1.728(4) |
| Al2a-016a | 1.722(4) | Al2b-O3bix | 1.725(4) | $\mathrm{Al2c-O3c}{ }^{\text {vi }}$ | 1.743(4) |
| Al2a-O3aiii | 1.732(4) | Al2b-O2a | 1.730(4) | Al2c-02b | 1.745(3) |
| Al2a-09a | 1.749(5) | Al2b-09b | 1.740(4) | $\mathrm{Al2c-O9c}$ | 1.746(4) |
| Mean | 1.730 | Mean | 1.729 | Mean | 1.741 |
| Al3a-O5b | 1.723(5) | Al3b-08a ${ }^{\text {viii }}$ | 1.707(4) | Al3c-07c | 1.737(4) |
| Al3a-O8c | 1.734(4) | Al3b-07bix | 1.724(4) | $\mathrm{Al3c-015c}$ | 1.745(3) |
| Al3a-015a | 1.739(4) | Al3b-O5c | 1.741(4) | Al3c-05a | 1.753(4) |
| Al3a-07a | 1.744(4) | Al3b-O15b | 1.748(4) | Al3c-O8b ${ }^{\text {vii }}$ | 1.755(3) |
| Mean | 1.735 | Mean | 1.730 | Mean | 1.747 |
| Al4a-O10c | 1.717(4) | Al4b-O1bix | 1.729(4) | Al4c-010b | 1.727(4) |
| Al4a-O14a | 1.730(5) | Al4b-O10a | 1.734(4) | Al4c-O14c | 1.740(4) |
| Al4a-O1a | 1.734(4) | Al4b-011b | 1.740(4) | Al4c-O1c | 1.745(3) |
| Al4a-011a | 1.739(4) | Al4b-O14b | 1.752(4) | Al4c-011c | 1.745(4) |
| Mean | 1.730 | Mean | 1.739 | Mean | 1.739 |
| Na1a-O14a | 2.399(2) | Na1b-O13c | 2.327(4) | $\mathrm{Na1c}-013{ }^{\text {viii }}$ | 2.321(4) |
| Na1a-O13b ${ }^{\text {x }}$ | 2.480(5) | Na1b-O14b | 2.514(5) | Na1c-O9c ${ }^{\text {viii }}$ | 2.392(4) |
| $\mathrm{Na1a-O15a}$ | 2.550(4) | Na1b-O9b | 2.519(4) | Na1c-O11c | 2.467(5) |
| Na1a-O10ax | 2.660(5) | Na1b-O11b | 2.572(4) | Na1c-O15c ${ }^{\text {viii }}$ | 2.477(4) |
| Na1a-O9a ${ }^{\text {vii }}$ | 2.687(2) | Na1b-O15b | 2.575(5) | Na1c-O12a | 2.654(6) |
| $\mathrm{Na1a-O16a}{ }^{\text {vii }}$ | 2.712(5) | Na1b-O16b | 2.653(5) | $\mathrm{Na1c}-\mathrm{O16c}$ viii | 2.973(5) |
| Na1a-O11a | 2.878(5) | Mean | 2.527 | Mean | 2.547 |
| Mean | 2.624 |  |  |  |  |
| $\mathrm{Na} 2 \mathrm{a}-\mathrm{O} 11 \mathrm{~b}^{\text {bx }}$ | 2.506(4) | $\mathrm{Na} 2 \mathrm{~b}-\mathrm{O} 6 \mathrm{~b}^{\text {vi }}$ | 2.403(4) | $\mathrm{Na} 2 \mathrm{c}-\mathrm{O} \mathrm{c}^{\text {vii }}$ | 2.362(4) |
| Na2a-O1a | 2.544(5) | $\mathrm{Na} 2 \mathrm{~b}-\mathrm{O} \mathrm{c}^{\text {vi }}$ | 2.428(4) | Na2c-O11a | 2.411(4) |
| $\mathrm{Na} 2 \mathrm{a}-\mathrm{O} 3 \mathrm{~b}^{\text {xi }}$ | 2.552(4) | $\mathrm{Na} 2 \mathrm{~b}-\mathrm{O} \mathrm{c}^{\text {vi }}$ | 2.474(4) | $\mathrm{Na} 2 \mathrm{c}-\mathrm{O} 3 \mathrm{a}$ | 2.516(4) |
| $\mathrm{Na} 2 \mathrm{a}-\mathrm{O7a}$ | 2.564(4) | Na2b-O16c | 2.486(5) | $\mathrm{Na} 2 \mathrm{c}-\mathrm{O} 5 \mathrm{a}$ | 2.690(4) |
| $\mathrm{Na} 2 \mathrm{a}-\mathrm{O16b}{ }^{\text {ix }}$ | 2.597(5) | $\mathrm{Na} 2 \mathrm{~b}-01 \mathrm{~b}^{\text {ix }}$ | 2.649(5) | $\mathrm{Na} 2 \mathrm{c}-\mathrm{O} 12 \mathrm{c}$ | 2.711(5) |
| Na2a-O6a | 2.744(5) | $\mathrm{Na} 2 \mathrm{~b}-\mathrm{O} 11 \mathrm{c}^{\text {vi }}$ | 2.791(4) | $\mathrm{Na2c}-016 \mathrm{a}^{\text {vii }}$ | 2.801(5) |
| $\mathrm{Na} 2 \mathrm{a}-\mathrm{O} 5 \mathrm{~b}$ | 2.909(2) | Mean | 2.539 | $\mathrm{Na} 2 \mathrm{c}-\mathrm{O7c}$ | 2.974(5) |
| Mean | 2.631 |  |  | Mean | 2.638 |
| Na3a-O2b | 2.357(4) | Na3b-O7b ${ }^{\text {xii }}$ | 2.403(5) | $\mathrm{Na3c}-07 \mathrm{c}^{\text {iii }}$ | 2.342 (4) |
| Na3a-O4a | 2.469(4) | $\mathrm{Na3b-O10}{ }^{\text {viii }}$ | 2.423(4) | $\mathrm{Na} 3 \mathrm{c}-\mathrm{O} 12 \mathrm{~b}{ }^{\text {ix }}$ | 2.399(4) |
| Na3a-O8c | 2.541(5) | Na3b-08a ${ }^{\text {ix }}$ | 2.460(4) | $\mathrm{Na3c}-\mathrm{O} 2 \mathrm{a}$ | 2.620(5) |
| Na3a-O14c | 2.542(5) | Na3b-O4b | 2.476(2) | $\mathrm{Na3c}-\mathrm{O} 4 \mathrm{c}$ | 2.716(4) |
| Na3a-O7a | 2.562(5) | Na3b-O2c | 2.563(5) | Na3c-O10b | 2.718(5) |
| Na3a-01a | 2.651(5) | Na3b-O1b ${ }^{\text {xi }}$ | 2.824(5) | $\mathrm{Na3c}-010 \mathrm{a}^{\text {iii }}$ | 2.735(5) |
| Mean | 2.520 | Na3b-O12a | 2.851(5) | Mean | 2.588 |
|  |  | Mean | 2.572 |  |  |
| Na4a-O6a | 2.541(6) | Na4b-O8b ${ }^{\text {xii }}$ | 2.373(5) | $\mathrm{Na4c}-\mathrm{O} \mathrm{c}^{\text {xiv }}$ | 2.434(5) |
| Na4a-O3b ${ }^{\text {xi }}$ | 2.693(5) | Na4b-O15b | 2.666(5) | Na4c-O8c | 2.576(5) |
| Na4a-O9a ${ }^{\text {vii }}$ | 2.714(3) | Na4b-O4c ${ }^{\text {vii }}$ | 2.720 (5) | Na4c-O5a ${ }^{\text {iii }}$ | 2.722(5) |
| $\mathrm{Na} 4 \mathrm{a}-\mathrm{O} \mathrm{b}^{\text {xiii }}$ | 2.886(3) | Na4b-O6c | 2.891(6) | Na4c-O8b | 2.829(6) |
| Na4a-O4b | 3.034(4) | $\mathrm{Na4b-O3c}{ }^{\text {vii }}$ | 3.051(7) | $\mathrm{Na} 4 \mathrm{c}-\mathrm{O} 2 \mathrm{~b}$ | 2.979(6) |
| Na4a-O2c ${ }^{\text {vii }}$ | 3.135(6) | Na4b-O2axiii | 3.078(7) | Na4c-O9c | 3.049(7) |
| Na4a-O8a ${ }^{\text {x }}$ | 3.158(6) | Na4b-O5c | 3.121(8) | $\mathrm{Na4c}-\mathrm{O} 15 \mathrm{c}^{\text {iii }}$ | 3.133(6) |
| $\mathrm{Na4a-O15axiil}$ | 3.171(6) | Mean | 2.843 | Mean | 2.817 |
| Mean | 2.916 |  |  |  |  |

Notes: Symmetry transformations used to generate equivalent atoms: (i) $-x+$ $1,-y+2, z-1 / 2$; (ii) $x, y, z-1$; (iii) $x-1, y, z$; (iv) $-x+1,-y+1, z-1 / 2 ;$ (v) $-x,-y+1$, $z-1 / 2 ;$ (vi) $x, y-1, z ;$ (vii) $x+1, y, z ;$ (viii) $x, y+1, z ;$ (ix) $-x+1,-y+1, z+1 / 2 ;$ (x) $-x$ $+2,-y+1, z+1 / 2 ;$ (xi) $x, y, z+1$; (xii) $x, y+1, z+1$; (xiii) $x+1, y+1, z ;$ (xiv) $x-1$, $y-1, z ;(x v)-x,-y+1, z+1 / 2 ;(x v i)-x+1,-y+2, z+1 / 2$.
graphical description of the structure. The reported value of $R_{1}$ was 0.117 , which is significantly higher than the value for our structure refinement.

As was mentioned in the introduction, several authors reported the existence of phase transitions occurring in nepheline-type materials at elevated temperatures. For the near future singlecrystal diffraction experiments are planned to investigate the high temperature behavior of monoclinic trinepheline in detail.


Figure 3. Three consecutive elliptical layers in the structure of hexagonal trinepheline (Kahlenberg and Böhm 1998) and their skeletal representation: (a) $0.00 \leq z \leq 0.24$; (b) $0.16 \leq z \leq 0.40$; (c) $0.32 \leq z \leq$ 0.56 .


Figure 4. The two layers building the unit cell in the structure of natural nepheline (Hahn and Buerger 1955), together with the skeletal representations: (a) $-0.05 \leq z \leq 0.55$; (b) $0.45 \leq z \leq 1.05$.

Table 6. Selected angles $\left({ }^{\circ}\right)$

| O3a-Si1a-O4a | 108.8(2) | O6bi-Si1b-O3b | 109.0(2) | O3c-Si1c-01c | 113.2(2) |
| :---: | :---: | :---: | :---: | :---: | :---: |
| O3a-Si1a-O6a | 109.9(2) | O6bi-Si1b-O1b | 107.8(2) | O3c-Si1c-O4c | 107.3(2) |
| O4a-Si1a-O6a | 111.1(2) | O3b-Si1b-01b | 111.9(2) | O1c-Si1c-O4c | 109.1(2) |
| O3a-Si1a-O1a | 111.3(2) | O6bi-Si1b-O4b ${ }^{\text {i }}$ | 111.8(2) | O3c-Si1c-O6ciil | 108.2(2) |
| O4a-Si1a-O1a | 108.5(2) | O3b-Si1b-O4b ${ }^{\text {i }}$ | 107.4(2) | O1c-Si1c-O6ciii | 110.6(2) |
| O6a-Si1a-O1a | 107.4(2) | O1b-Si1b-O4b ${ }^{\text {i }}$ | 109.0(2) | O4c-Si1c-O6ciil | 108.3(2) |
| Mean | 109.5 | Mean | 109.5 | Mean | 109.4 |
| O14aii-Si2a-O7b | 108.8(2) | O14b-Si2b-O16c | 109.8(2) | O14c-Si2c-O12a | 109.1(3) |
| O14a-Si2a-012b | 108.0(2) | O14b-Si2b-O7c | 109.3(2) | O14c-Si2c-016a | 110.8(2) |
| O7b-Si2a-O12b | 112.1(2) | O16c-Si2b-07c | 108.7(2) | O12a-Si2c-016a | 111.2(2) |
| O14ai'-Si2a-O16b ${ }^{\text {iv }}$ | 110.2(2) | O14b-Si2b-O12c | 109.0(2) | O14c-Si2c-07a | 106.2(2) |
| O7b-Si2a-O16biv | 108.7(2) | O16c-Si2b-O12c | 110.8(2) | O12a-Si2c-07a | 109.8(2) |
| O12b-Si2a-O16biv | 109.0(2) | O7c-Si2b-O12c | 109.2(2) | O16a-Si2c-07a | 109.6(2) |
| Mean | 109.5 | Mean | 109.5 | Mean | 109.4 |
| O9ai-Si3a-08a ${ }^{\text {iv }}$ | 108.4(2) | O10b-Si3b-O2b | 112.3(2) | O10c-Si3c-O9c | 109.3(2) |
| O9ai-Si3a-02av | 110.5(2) | O10b-Si3b-O9b | 108.5(2) | O10c-Si3c-O2c ${ }^{\text {vi }}$ | 106.4(2) |
| O8a'*-Si3a-02av | 109.4(2) | O2b-Si3b-O9b | 108.7(2) | O9c-Si3c-O2c ${ }^{\text {vi }}$ | 113.3(2) |
| O9ai-Si3a-O10aiv | 108.8(2) | O10b-Si3b-O8b | 108.9(2) | O10c-Si3c-O8c | 114.0(2) |
| O8a ${ }^{\text {w- }}$-Si3a-O10a ${ }^{\text {iv }}$ | 111.7(2) | O2b-Si3b-08b | 108.9(2) | O9c-Si3c-08c | 106.0(2) |
| O2av-Si3a-O10aiv | 108.0(2) | O9b-Si3b-O8b | 109.5(2) | O2c ${ }^{\text {vi-Si3c-O8c }}$ | 108.1(2) |
| Mean | 109.5 | Mean | 109.5 | Mean | 109.5 |
| O13a-Si4a-O11a | 109.1(2) | O13b-Si4b-O15b | 111.4(2) | O11c-Si4c-O15c ${ }^{\text {viii }}$ | 107.8(2) |
| O13a-Si4a-O15a ${ }^{\text {vii }}$ | 109.6(2) | O13b-Si4b-O5b ${ }^{\text {iv }}$ | 108.8(2) | O11c-Si4c-O5c | 108.9(2) |
| O11a-Si4a-O15a ${ }^{\text {vii }}$ | 109.7(2) | O15b-Si4b-O5b ${ }^{\text {iv }}$ | 109.0(2) | O15c ${ }^{\text {viï-Si4c-O5c }}$ | 109.0(2) |
| O13a-Si4a-O5a | 109.2(2) | 013b-Si4b-011b | 110.0(2) | O11c-Si4c-O13c | 111.3(2) |
| O11a-Si4a-O5a | 109.8(2) | O15b-Si4b-011b | 109.1(2) | O15c ${ }_{\text {viii-Si4c-O13c }}$ | 109.9(2) |
| O15a ${ }^{\text {vii-Si4a-O5a }}$ | 109.5(2) | O5b ${ }^{\text {iv-Si4b-011b }}$ | 108.5(2) | O5c-Si4c-O13c | 109.9(2) |
| Mean | 109.5 | Mean | 109.5 | Mean | 109.5 |
| O12a-Al1a-O13a ${ }^{\text {viii }}$ | 105.0(2) | O6b-Al1b-O4c | 106.8(2) | O12c-Al1c-O4a | 105.5(2) |
| O12a-Al1a-O4b | 108.1(2) | O6b-Al1b-O13biii | 111.6(2) | O12c-Al1c-O13c | 112.7(2) |
| O13aviil-Al1a-O4b | 111.7(2) | O4c-Al1b-013biii | 110.0(2) | O4a-Al1c-O13c | 109.3(2) |
| O12a-Al1a-O6a | 114.7(2) | O6b-Al1b-O12b ${ }^{\text {ix }}$ | 111.8(2) | O12c-Al1c-O6c | 109.9(2) |
| O13a ${ }^{\text {viil-Al1a-O6a }}$ | 110.0(2) | O4c-Al1b-012bix | 105.9(2) | O4a-Al1c-O6c | 111.1(2) |
| O4b-Al1a-O6a | 107.4(2) | O13b ${ }^{\text {ii- }}$ Al1b-O12 ${ }^{\text {bx }}$ | 110.5(2) | O13c-Al1c-O6c | 108.3(2) |
| Mean | 109.5 | Mean | 109.4 | Mean | 109.5 |
| O2c-Al2a-O16a | 113.1(2) | O16b-Al2b-O3b ${ }^{\text {ix }}$ | 106.4(2) | O16c-Al2c-O3c ${ }^{\text {vi }}$ | 106.9(2) |
| O2c-Al2a-O3aii | 109.6(2) | O16b-Al2b-O2a | 113.3(2) | O16c-Al2c-O2b | 111.0(2) |
| O16a-Al2a-O3aii | 107.9(2) | O3bix-Al2b-O2a | 112.1(2) | O3c ${ }^{\text {vi-Al2c-O2b }}$ | 111.3(2) |
| O2c-Al2a-09a | 107.2(2) | O16b-Al2b-09b | 107.4(2) | O16c-Al2c-09c | 109.3(2) |
| O16a-Al2a-09a | 109.4(2) | O3bix-Al2b-O9b | 110.2(2) | O3c ${ }^{\text {vi-Al2c-O9c }}$ | 107.0(2) |
| O3aii-Al2a-O9a | 109.6(2) | O2a-Al2b-O9b | 107.3(2) | O2b-Al2c-O9c | 111.1(2) |
| Mean | 109.5 | Mean | 109.4 | Mean | 109.4 |
| O5b-Al3a-O8c | 108.9(2) | O8a ${ }^{\text {viil-Al3b-07b }}$ ix | 103.5(2) | O7c-Al3c-O15c | 115.7(2) |
| O5b-Al3a-015a | 107.4(2) | O8a ${ }^{\text {viii-Al3b-O5c }}$ | 112.6(2) | O7c-Al3c-05a | 109.9(2) |
| O8c-Al3a-015a | 111.5(2) | O7bix-Al3b-O5c | 109.4(2) | O15c-Al3c-O5a | 106.5(2) |
| O5b-Al3a-07a | 108.6(2) | O8aviil-Al3b-015b | 110.9(2) | O7c-Al3c-O8b ${ }^{\text {vii }}$ | 110.8(2) |
| O8c-Al3a-07a | 107.8(2) | O7bix-Al3b-015b | 116.8(2) | O15c-Al3c-O8b ${ }^{\text {vi }}$ | 106.0(2) |
| O15a-Al3a-07a | 112.5(2) | O5c-Al3b-O15b | 104.0(2) | O5a-Al3c-O8b ${ }^{\text {vii }}$ | 107.5(2) |
| Mean | 109.4 | Mean | 109.5 | Mean | 109.4 |
| O10c-Al4a-O14a | 109.8(2) | O1b ${ }^{\text {ix }-A l 4 b-O 10 a ~}$ | 110.3(2) | O10b-Al4c-O14c | 109.3(2) |
| O10c-Al4a-O1a | 106.7(2) | O1b ${ }^{\text {ix-Al4b-011b }}$ | 109.7(2) | O10b-Al4c-O1c | 107.5(2) |
| O14a-Al4a-O1a | 110.5(2) | O10a-Al4b-011b | 110.3(2) | O14c-Al4c-O1c | 109.7(2) |
| O10c-Al4a-O11a | 112.8(2) | O1b ${ }^{\text {ix-Al4b-O14b }}$ | 110.2(2) | O10b-Al4c-O11c | 113.2(2) |
| O14a-Al4a-O11a | 106.1(2) | O10a-Al4b-014b | 111.7(2) | O14c-Al4c-O11c | 109.2(2) |
| O1a-Al4a-O11a | 110.9(2) | O11b-Al4b-O14b | 104.5(2) | O1c-Al4c-O11c | 107.9(2) |
| Mean | 109.5 | Mean | 109.4 | Mean | 109.5 |
| Si1a-O1a-Al4a | 135.9(2) | $\mathrm{Si3} \mathrm{a}^{\text {av }}$-O2a-Al2b | 138.4(3) | Si1a-O3a-Al2avii | 138.0(3) |
| Si1b-O1b-Al4b ${ }^{\text {iv }}$ | 144.2(3) | Si3b-O2b-Al2c | 130.0(2) | Si1b-O3b-Al2b ${ }^{\text {iv }}$ | 138.2(3) |
| Si1c-O1c-Al4c | 137.6(2) | Si3c ${ }^{\text {viii-O2 }}$ c-Al2a | 150.8(3) | Si1c-O3c-Al2c ${ }^{\text {viii }}$ | 141.6(3) |
| Si1a-O4a-Al1c | 132.6(2) | Si4a-O5a-Al3c | 137.7(2) | Si1a-O6a-Al1a | 135.9(2) |
| Si1bxi-O4b-Al1a | 134.5(1) | Si4bix-O5b-Al3a | 132.5(1) | Si1b ${ }^{\text {xvi }}$-O6b-Al1b | 140.8(2) |
| Si1c-O4c-Al1b | 139.8(2) | Si4c-O5c-Al3b | 140.8(2) | Si1c ${ }^{\text {vii-O6c-Al1c }}$ | 132.4(2) |
| Si2c-O7a-Al3a | 136.7(2) | Si3a ${ }^{\text {ix-0 }}$-08a-Al3b ${ }^{\text {vi }}$ | 148.4(3) | Si3axio9a-Al2a | 139.5(1) |
| Si2a-O7b-Al3biv | 142.9(2) | Si3b-O8b-Al3ciii | 131.4(2) | Si3b-09b-Al2b | 134.0(2) |
| Si2b-O7c-Al3c | 137.6(2) | Si3c-O8c-Al3a | 133.9(2) | Si3c-09c-Al2c | 139.1(2) |
| Si3a-O10a-Al4b | 138.5(2) | Si4a-O11a-Al4a | 137.5(2) | Si2c-012a-Al1a | 145.9(3) |
| Si3b-O10b-Al4c | 156.03) | Si4b-011b-Al4b | 134.8(2) | Si2a-O12b-Al1b ${ }^{\text {iv }}$ | 135.4(2) |
| Si3c-O10c-Al4a | 146.6(3) | Si4c-011c-Al4c | 138.9(2) | Si2b-O12c-Al1c | 152.1(3) |
| Si4a-O13a-Al1a ${ }^{\text {vi }}$ | 139.9(2) | Si2a ${ }^{\text {xi }}$-O14a-Al4a | 147.4(1) | Si4aiio-O15a-Al3a | 140.4(2) |
| Si4b-O13b-Al1 b ${ }^{\text {vii }}$ | 140.8(3) | Si2b-O14b-Al4b | 153.5(3) | Si4b-O15b-Al3b | 132.2(2) |
| Si4c-013c-Al1c | 128.9(2) | Si2c-014c-Al4c | 157.0(3) | Si4c ${ }^{\text {vi- }}$-15c-Al3c | 137.9(2) |
| Si2c-016a-Al2a | 139.6(2) |  |  |  |  |
| Si2ax-016b-Al2b | 136.6(2) |  |  |  |  |
| Si2b-016c-Al2c | 139.7(2) |  |  |  |  |

Notes: Symmetry transformations used to generate equivalent atoms: (i) $-x+1,-y+2, z-1 / 2$; (ii) $x, y, z-1$; (iii) $x-1, y, z$; (iv) $-x+1,-y+1, z-1 / 2$; (v) $-x,-y+1, z-1 / 2$; (vi) $x, y-1, z ;$ (vii) $x+1, y, z$; (viii) $x, y+1, z ;$ (ix) $-x+1,-y+1, z+1 / 2 ;$ (x) $-x+2,-y+1, z+1 / 2 ;$ (xi) $x, y, z+1$; (xii) $x, y+1, z+1$; (xiii) $x+1, y+1, z ;($ (xiv) $x-1, y-1, z ;(x v)$ $-x,-y+1, z+1 / 2 ;(x v i)-x+1,-y+2, z+1 / 2$.

TABLE 7. Bond-valence ranges for cations and bond-valence sums for all atoms in the structure of the monoclinic trinepheline

| Atom | B. V. S | B. V. range | Atom | B. V. S | B. V. range | Atom | B. V. S | B.V. range |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Si1a | 4.02 | 0.97-1.03 | Al1a | 3.23 | 0.78-0.83 | Na1a | 0.82 | 0.05-0.20 |
| Si1b | 4.10 | 1.00-1.05 | Al1b | 3.18 | 0.78-0.83 | Na1b | 0.88 | 0.10-0.24 |
| Si1c | 4.09 | 1.00-1.06 | Al1c | 3.13 | 0.77-0.80 | Na1c | 0.91 | 0.04-0.25 |
| Si2a | 4.00 | 0.98-1.02 | Al2a | 3.24 | 0.77-0.84 | Na 2 a | 0.78 | 0.05-0.15 |
| Si2b | 4.16 | 1.01-1.09 | Al2b | 3.24 | 0.78-0.83 | Na 2 b | 0.87 | 0.07-0.20 |
| Si2c | 4.15 | 0.99-1.08 | Al2c | 3.14 | 0.77-0.81 | Na 2 c | 0.84 | 0.04-0.22 |
| Si3a | 4.06 | 1.01-1.03 | Al3a | 3.19 | 0.78-0.83 | Na3a | 0.88 | 0.10-0.22 |
| Si3b | 4.02 | 0.99-1.04 | Al3b | 3.24 | 0.77-0.86 | Na3b | 0.96 | 0.06-0.20 |
| Si3c | 4.12 | 1.00-1.04 | Al3c | 3.08 | 0.76-0.79 | Na3c | 0.79 | 0.08-0.23 |
| Si4a | 4.06 | 1.01-1.03 | Al4a | 3.23 | 0.79-0.84 | Na 4 a | 0.48 | 0.03-0.14 |
| Si4b | 4.05 | 0.99-1.08 | Al4b | 3.16 | 0.76-0.81 | Na4b | 0.54 | 0.03-0.21 |
| Si4c | 4.05 | 0.98-1.03 | Al4c | 3.15 | 0.77-0.81 | Na4c | 0.55 | 0.03-0.18 |
| O1a | -2.00 |  | O6b | -2.07 |  | O11c | -2.04 |  |
| 01b | -1.99 |  | O6c | -2.00 |  | 012a | -2.02 |  |
| O1c | -2.01 |  | 07a | -2.02 |  | 012b | -1.97 |  |
| O2a | -1.95 |  | 07b | -2.04 |  | O12c | -1.90 |  |
| O2b | -2.04 |  | O7c | -2.09 |  | 013a | -2.10 |  |
| O2c | -2.03 |  | O8a | -2.07 |  | 013b | -2.02 |  |
| O3a | -1.98 |  | 08b | -2.02 |  | O13c | -1.99 |  |
| O3b | -2.07 |  | O8c | -2.05 |  | 014a | -2.03 |  |
| O3c | -2.04 |  | 09a | -1.98 |  | 014b | -2.00 |  |
| O4a | -1.96 |  | O9b | -1.92 |  | O14c | -2.00 |  |
| O4b | -2.00 |  | O9c | -2.05 |  | 015a | -1.96 |  |
| O4c | -1.97 |  | O10a | -1.98 |  | O15b | -1.98 |  |
| O5a | -1.94 |  | O10b | -1.94 |  | O15c | -1.99 |  |
| O5b | -1.92 |  | O10c | -2.06 |  | 016a | -2.01 |  |
| O5c | -2.01 |  | 011a | -2.05 |  | 016b | -2.02 |  |
| O6a | -1.99 |  | 011b | -2.05 |  | O16c | -2.04 |  |

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## REFERENCES CITED

Balić-Žunić, T. and Vicković, I. (1996) IVTON—A program for the calculation of geometrical aspects of crystal structures and some crystal chemical applications. Journal of Applied Crystallography, 29, 305-306.
Brown, I.D. (1981) The bond-valence method: an empirical approach to crystal structure and bonding. In M. O’Keeffe and A. Navrotsky, Eds., Structure and Bonding in Crystals II, p. 1-30. Academic Press, New York.
Brown, W.L., Cesbron, F., and Dupont, G. (1972) Trinepheline; a new synthetic modification in the nepheline group. Zeitschrift für Kristallographie, 136, 468-470.
Burla, M.C., Caliandro, R., Camalli, M., Carrozzini, B., Cascarano, G.L., De Caro, L., Giacovazzo, C., Polidori, G., and Spagna, R. (2005) SIR2004: An improved tool for crystal structure determination and refinement. Journal of Applied Crystallography, 38, 381-388.
Busing, W.R. and Levy, H.A. (1964) The effect of thermal motion on the estimation of bond lengths from diffraction measurements. Acta Crystallographica, 17, 142-146.
Crystal Impact (2005) DIAMOND 3.1a, Visual Crystal Structure Information System. Crystal Impact, Bonn, Germany.
Dimitrijević, R., Dondur, V., Vulić, P., Marković, S., and Macura, S. (2004) Structural characterization of pure Na-nephelines synthesized by zeolite conversion route. Journal of Physics and Chemistry of Solids, 65, 1623-1633.
Dollase, W.A. and Thomas, W.M. (1978) The crystal chemistry of silica rich, alkalideficient nepheline. Contributions to Mineralogy and Petrology, 66, 311-318.
Fischer, R.X., LeLirzin, A., Kassner, D., and Rüdinger, B. (1991) STRUPLO91, eine neue Version des Fortranprogramms zur Darstellung von Kristallstrukturen. Zeitschrift für Kristallographie, Supplemental Issue, 375.
Flörke, O.W. (1967) Die Modifikationen des $\mathrm{SiO}_{2}$. Fortschritte der Mineralogie, 44, 181-230.
Gregorkiewitz, M. (1984) Crystal structure and Al/Si-ordering of a synthetic nepheline. Bulletin Minèralogique, 107, 499-507.
Hahn, T. and Buerger, M.J. (1955) The detailed structure of nepheline $\mathrm{KNa}_{3} \mathrm{Al}_{4} \mathrm{Si}_{4} \mathrm{O}_{16}$. Zeitschrift für Kristallographie, 106, 308-338.
Henderson, C.M.B. and Roux, J. (1977) Inversion in sub-potassic nephelines. Contributions to Mineralogy and Petrology, 61, 279-298.
Henderson, C.M.B. and Thompson, A.B. (1980) The low-temperature inversion in sub-potassic nephelines. American Mineralogist, 65, 970-980.
Hippler, B. and Böhm, H. (1989) Structure investigation on sodium nephelines.

Zeitschrift für Kristallographie, 187, 39-53.
Hovestreydt, E. (1983) On the atomic scattering factor for $\mathrm{O}^{2-}$. Acta Crystallographica, A39, 268-269.
Jarchow, O., Reese, H.H., and Saalfeld, H. (1966) Hydrothermalsynthesen von Zeolithen der Sodalith-und Cancrinitgruppe. Neues Jahrbuch für Mineralogie Monatshefte, 10, 289-297.
Jones, J.B. (1968) Al-O and Si-O tetrahedral distances in aluminosilicate framework structures. Acta Crystallographica, B24, 355-358.
Kahlenberg, V. and Böhm, H. (1998) Crystal structure of hexagonal trinepheline-A new synthetic $\mathrm{NaAlSiO}_{4}$ modification. American Mineralogist, 83, 631-637.
Klaska, K.H. (1974) Strukturuntersuchungen an Tridymitabkömmlingen. Ph.D. thesis, Fachbereich Geowissenschaften, Univesität Hamburg, Germany.
Liebau, F. (1985) Structural chemistry of silicates. Springer Verlag, Berlin.
Loewenstein, W. (1954) The distribution of Al in tetrahedra of silicates and aluminates. American Mineralogist, 39, 92-96.
Pluth, J.J. and Smith, J.V. (1980) Accurate redetermination of crystal structure of dehydrated zeolite A. Absence of near zero coordination of sodium. Refinement of Si, Al-ordered superstructure. Journal of the American Chemical Society, 102, 4704-4708.
Prince, E. (2004) International Tables for Crystallography, vol. C. Dordrecht, The Netherlands.
Robinson, K., Gibbs, G.V., and Ribbe, P.H. (1971) Quadratic elongation: A quantitative measure of distortion in polyhedra. Science, 171, 567-570.
Roth, G. (1985) Zum Zusammenhang zwischen Ionenleitung und Kristallstruktur von Festkörperelektrolyten: Theoretische Modelle sowie experimentelle Untersuchungen an Phasen des Nephelin Typs. Ph.D. thesis, Universität Münster, Germany.
Schneider, H., Flörke, O.W., and Stoeck, R. (1994) The $\mathrm{NaAlSiO}_{4}$ nepheline-carnegieite solid-state transformation. Zeitschrift für Kristallographie, 209, 113-117.
Selker, P. (1987) Hydrothermalsynthesen und Strukturuntersuchungen von zwei Trinephelin-Modifikationen der Zusammensetzung $\mathrm{NaAlSiO}_{4}$. Ph.D. thesis, Ruhr-Universität Bochum, Germany.
Selker, P., Bartsch, H.H., and Klaska, R. (1985) Struktur und Hydrothermalsynthesen von $\mathrm{NaAlSiO}_{4}$ Modifikationen. Zeitschrift für Kristallographie, 170, 175-176.
Shannon, R.D. and Prewitt, C.T. (1969) Effective ionic radii in oxides and fluorides. Acta Crystallographica, B25, 925-946.
Sheldrick, G.M. (1998) SHELXL97—Program for crystal structure analysis (Release 97-2). Universität Göttingen, Germany.
Stoe and Cie GmbH (2005) X-Area program. Darmstadt, Germany.
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